

Article A Study on the Conductivity and Infrared Spectroscopy of Elastic Polymer Composite Materials

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Abstract: Elastic polymer composite materials possess physical properties such as softness, durability, and heat preservation, making them suitable for designing flexible, snug, and highly ductile devices. While polymers are generally electrical insulators, they can become electrically conductive when doped with carbon powder. Carbon can strongly interact with polymers, enabling the composites to exhibit electrical conductivity through this adsorption process. By heating and melting the polymer and incorporating thin but heavily concentrated carbon powder, these composites can achieve electrical properties similar to those of conductors or semiconductors. This study examines the adsorption and electrical properties of elastic polymer composite materials, focusing on aspects such as resistance temperature characteristics and conductivity. The results indicate that electrically conductive elastic polymers have potential applications in electrothermal devices, thermostatic control systems, sensors, and variable resistors. Additionally, scanning electron microscopy (SEM) is employed to analyze the material's structure for further research.

Keywords: elastic polymer composite materials; adsorption; carbon powder; electrical property; scanning electron microscope; infrared spectroscopy

1. Introduction

The advantages of using carbon in electrical applications are well documented and include low cost, a broad potential window, relatively inert electrochemistry, and electrocatalytic activity for various redox reactions [1]. Several reviews and monographs have highlighted the use of carbon as an electronic material, focusing particularly on classical carbon materials such as graphite, glassy carbon, and carbon black [2–4]. Additionally, mesoporous carbon, a high-surface-area graphitic material, is produced through the "carbonization" or pyrolysis of carbon-rich precursors like sucrose [5,6].

Carbon polymers exhibit a broad range of conductivities due to the diverse nature of carbon materials, making them suitable for electronic applications. In this study, carbon particles are utilized as the foundational electronic components, leveraging the properties of high-surface-area carbon materials. This technique has the potential to be widely applied in developing new circuit components and advancing electronic technologies and processes.

1.1. Physisorption and Conductivity

When heating elastic polymer composite materials doped with carbon or activated carbon powder, the low-melting-point polymer evaporates completely, leaving behind the carbon, which maintains its mass and size. This process involves physisorption, also known as van der Waals adsorption, rather than chemisorption, as it does not involve any chemical reactions between the carbon and the polymer.

1.2. Adsorptive Reaction of Elastic Polymer Composite Material and Carbon Powder

Adsorption refers to the process where a material interface attracts and temporarily holds medium particles from its surroundings. Materials that can adsorb particles on their



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Copyright: © 2024 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). This dual adsorption creates opposing forces, which can reduce the surface tension of the carbon. As a result, the interaction between the carbon adsorbent and the polymer composite material is enhanced, potentially improving the composite's properties and performance.

However, the elastic polymer composite material's molecules adsorbed on carbon's surface are not static because the heated perturbation or elastic collisions of these molecules can cause them to deviate from the surface of the adsorbent and return to the surrounding media [7,8]. This phenomenon is called desorption. Adsorption and desorption are reversible reactions; physisorption also has the property of surface adsorbent saturation. At a fixed temperature, adsorption can reach equilibrium with the reaction time, which is named adsorptive equilibrium. Adsorptive equilibrium can be acquired by the basic dynamic model and the statistical mechanics method. The brief expression is shown as

Free molecule $\stackrel{\text{Adsorption}}{\underset{\text{Desorption}}{\Rightarrow}}$ Molecule being adsorbed + Head of adsorption (1)

2. Materials and Methods

and the carbon molecules.

2.1. The Base Material of Elastic Polymer Composite Materials

Using the soft, insulating, and durable properties of elastic polymer materials, we design devices that fit closely and provide flexibility and resilience. However, these elastic polymers are typically non-conductive. Through research and experimentation, it was discovered that polysiloxy polymers, which are often silicones, exhibit strong adsorption properties.

Polysiloxy polymers are characterized by their silicon–oxygen backbone, with the basic repeating unit being –Si–O–. To impart electrical conductivity, these polymers are doped with a high concentration of graphite particles. Silicones are known for their stable, non-toxic structure and rubber-like physical properties, making them an ideal base material when combined with conductive fillers like graphite. The structural unit of silicones is shown in Figure 1a.



Figure 1. Repetitive structural units of (a) polysiloxy polymers; (b) polyurethane.

Silicon atoms in polysiloxy polymers are bonded with various organic functional groups, such as methyl, ethyl, and phenyl, making these polymers semi-organic and semi-inorganic. Oxygen plasma treatment can enhance the $-SiO_n(OH)_{4-n}$ groups on the silicon surface, with the degree of modification being influenced by the intensity of the oxygen plasma and the treatment duration [9].

To investigate the electrical properties related to carbon doping concentrations, different doping levels of the polymeric rubber-like material were produced. Highly adsorptive carbon was chosen for doping within silicon-based polymers and polyurethane rubber. The repetitive structural units of these materials are depicted in Figure 1b as thermoplastic rubber (TPR). This approach aimed to explore how varying concentrations of carbon influence the electrical properties of the composite material.

2.2. The Analysis of SEM

The molecular packing of polymers significantly impacts their physical properties. Carbon physisorption reduces the intermolecular distance while substantially increasing intermolecular attraction. The adsorption behavior at different doping concentrations in elastic polymer composite materials was observed through SEM, as shown in Figure 2. These images highlight the secondary bonding forces within the polymer and the high residual forces due to macromolecular weight. When the carbon dopant concentration is too high, it creates a strong attraction, leading to tightly packed molecules arranged in a highly regular structure. However, due to the polymer's macromolecular weight and structural defects, perfect parallel alignment is not achievable.



Figure 2. SEM images of elastic polymer composite materials with varying levels of carbon powder doping: (**a**) 0 M; (**b**) 4 M; (**c**) 20 M; (**d**) 40 M; (**e**) 80 M.

Carbon atoms are composed of three sp² hybridized orbitals, which interact when in close proximity. In a graphite structure, carbon atoms have an electronic configuration of 2S-2P. Each layer of carbon atoms forms strong σ bonds with three adjacent carbon atoms. The bonding in the vertical direction between layers is facilitated by unhybridized P orbitals forming π bonds, which are relatively weak, resulting in a very narrow energy gap. This narrow gap allows valence electrons to move easily within the π orbitals, providing good electrical conductivity. SEM images from Figure 2a–e show that increasing the concentration of carbon powder enhances the physisorption function. When the distance between carbon particles approaches the van der Waals radius (0.3354 ± 0.0001 nm), valence electrons begin to overlap intermolecularly, significantly increasing electrical conductivity.

2.3. Doping Concentration of Carbon and Electrical Conductivities

When enough carbon atoms are close to each other as shown on Figure 3, they will form a continuous permissible energy level. This energy level can cause the electrons which were fixed on the valence band to move interatomically. It is not possible to distinguish which atoms these electrons belong to, which is called electron sharing. These electrons have both around atomic and communization movement characteristics, so they will split into the energy levels of many groups. The energy levels of each group are very close, so the energy levels turn into approximate continuity, while the composition of a particular band width, with this kind of energy level arranged as close-order belt shape, is called the energy band.



Figure 3. The electronic structure of carbon atoms [10].

An important consideration regarding electrode materials is the concentration of electronic states (DOS), which varies greatly for different forms of carbon. The high conductivity of metals results from the combination of a large number of atomic orbits to form bands with a high concentration of electronic states.

Sole atoms that only have one energy level are those where the electron is bound on the energy level. When another atom is very close, the atoms interact and the energy level splits into two, resulting in a degenerated state. If there is one more atom approaching, it will split into three energy levels. As N's isolated atoms are brought together to form a solid, they will split into N's energy levels and become very close to each other. The electronic structure of each individual carbon atom is $1S^22S^22P^4$, so a single carbon atom has two 1S states, two 2S states, four 2P states and higher energy state. Ns' carbon atoms will have $2 \times Ns'$ 1S states, $2 \times Ns'$ 2S states and $4 \times Ns'$ 2P states.

Figure 4 is a solid model of the energy gap; the region between different energy levels without any electronics is called the energy gap or forbidden band. The width of the energy gap is the indicator of the difference between the insulator, semiconductor, and conductor. The carbon atoms of graphite's structure have very good electric conductivity because there is not an energy gap between the valence band and conduction band. Electrons move freely without changing energy, and metallic conduction follows Ohm's law [11].



Figure 4. Solid energy gap model.

When the distance between carbon atoms is smaller, then the carbon atoms will form a diamond lattice as Figure 5 shows. The 2S-2P energy level can be dispersed to two energy levels and possess $8 \times \text{Ns'}$ states. The top energy level is called the conduction band which has $4 \times \text{Ns'}$ states, and the bottom energy level is called the valence band, which also has $4 \times \text{Ns'}$ states. These two energy levels are separated by a very wide energy gap, a large band gap of more than 5 eV, so undoped diamond is normally electrically insulating and cannot be used as an electrode material [12]. But as with other large-band-gap materials, diamond can be made conductive by doping it with certain elements [13].

Diamond electrodes that make use of the surface transfer doping effect have been named surface conductive diamond (SCD) materials [14]. The electroanalytical techniques for which diamond electrodes have been applied are called amperometric detection [15–35].



Figure 5. Diamond lattice, for example, Si, Ge, C..., etc. (**A**) Tetrahedron bond in diamond lattice. (**B**) A tetrahedron bond. (**C**) Schematic two-dimensional representation of a tetrahedron bond [36]. Explanation: The points in Figure (**C**) depict the electrons shared within the covalent bond.

2.4. Infrared Absorption Spectrum Analysis

Figure 6 presents the infrared absorption spectrum of the base material, which exhibits four main absorption peaks at the following wavenumbers: 1030 cm⁻¹, 1600 cm⁻¹, 2260 cm⁻¹, and 2315 cm⁻¹. Figures 7 and 8 illustrate the infrared spectra of elastic polymer composite materials doped with varying concentrations of carbon powder, showing how these peaks shift or change with different doping levels.



Figure 6. The infrared spectrum illustration of the base material. Elastic polymer composite materials without carbon powder doping.



Figure 7. The infrared spectrum illustration of elastic polymer composite materials doped with 4 M carbon powder.

As the doping concentration of carbon powder increases, the absorbance of lower energy chemical bonds decreases significantly. Consequently, absorption peaks at wavenumbers less than 1500 cm^{-1} nearly vanish. The characteristic infrared spectra of the elastic polymer composite materials in this study are primarily attributed to various vibrational modes of chemical bonds, as explained in the following.

7 of 16



Figure 8. The infrared spectrum illustration of elastic polymer composite materials doped with 40 M carbon powder.

2.4.1. The Infrared Spectrum Illustration of C-O Group

The absorption peak corresponding to the C–O group stretching vibration, which is found at 1030 cm⁻¹, diminishes as the concentration of carbon particles in the substrate increases. This indicates an inverse relationship between the C–O group concentration and the carbon particle doping concentration. As the doping concentration of carbon powder rises, the concentration of C–O groups in the base material decreases.

2.4.2. The Infrared Spectrum Illustration of C=C Group

In the characteristic region of the C=C group stretching vibration, the absorption spectrum typically features a single peak near the wavenumber of 1600 cm⁻¹ (6.25 μ m). The intensity of this peak is directly proportional to the doping concentration of carbon particles in the substrate. As the concentration of carbon powder increases, the number of C=C groups in the base material also rises, leading to a more pronounced absorption peak in this region.

2.4.3. The Infrared Spectrum Illustration of C≡C Group

At low doping concentrations of carbon powder in the substrate, the absorption spectrum displays a single, reduced absorption peak located at 2260 cm⁻¹. However, at higher doping concentrations, the spectrum reveals two distinct absorption peaks at wavenumbers of 2260 cm⁻¹ and 2315 cm⁻¹, respectively.

2.5. UV/VIS/NIR Spectroscopy

Figure 9 illustrates the UV-VIS-NIR spectroscopy of the base material with varying concentrations of carbon powder doping. The UV-VIS-NIR absorption spectrum reveals a consistent pattern across different doping concentrations, with a common absorption peak at 325 nm. This peak is attributed to the transition of electrons from the valence band to the conduction band within the polymer chain of the elastic polymer composite materials [37]. As the concentration of carbon powder increases, a stronger absorption peak emerges at a wavelength of 2310 nm.



Figure 9. The UV-VIS-NIR absorption spectroscopy of elastic polymer composite materials with different carbon powder doping concentrations.

UV-VIS-NIR absorption spectroscopy provides insights into chemical bonding, particularly in relation to the electronic structures of polymers, including electronic orbitals, valence shells, and molecular configurations [38–42]. When the doping concentration of carbon powder in the base material is increased to 40 M, the conductive elastic polymer composite material exhibits a more pronounced absorption peak at 2310 nm. This enhancement occurs because the valence electrons, initially confined to the valence band, transition to the conduction band as the doping concentration rises. The carbon powder's physisorption contributes to the excitation of these electrons, turning them into free carriers. Consequently, the material's conductivity improves significantly, shifting the absorption peak into the near-infrared region, between 1500 nm and 2500 nm.

Figure 9 reveals how the material's absorption properties change with varying levels of doping. As the concentration of carbon powder increases, the absorption spectrum displays notable shifts and intensifications. Specifically, the presence of carbon powder enhances the absorption peaks, with significant changes occurring at wavelengths that correspond to the increased electronic transitions and improved conductivity in the composite materials. This effect causes the absorption peaks to shift towards the near-infrared region as the doping concentration rises.

2.6. EDX Analysis

This study utilized energy-dispersive X-ray (EDX) analysis to investigate the composition and content of elastic polymer composite material specimens. Figure 10 and Table 1 present the results of the EDX analysis. The findings indicate that the elastic polymer composite material contains 4.9% silicon after doping with 40 M graphite, suggesting that the silicon atomic structure significantly influences the adsorption and conductivity of the elastic polymer composite material.



Figure 10. EDX images of elastic polymer composite materials: (**a**) Composite with 0 M graphite doping. Elastic polymer composite materials without carbon powder doping. (**b**) Composite with 40 M graphite doping.

| Sample | Α | | В | |
|---------|---------|---------|---------|---------|
| Element | Weight% | Atomic% | Weight% | Atomic% |
| С | | | 70.39 | 80.68 |
| 0 | 39.41 | 56.57 | 14.83 | 12.76 |
| Si | 14.70 | 12.02 | 4.90 | 2.40 |
| Al | 8.24 | 7.01 | 2.59 | 1.32 |
| Cl | 37.65 | 24.39 | 7.28 | 2.83 |
| Total | 100 | | 100 | |
| | | | | |

Table 1. The EDX analysis result. (A) Elastic polymer composite materials doped with 0 M graphite specimen. (B) Elastic polymer composite materials doped with 40 M graphite specimen.

3. Results

3.1. Activated Carbon's Characteristic Equation

The relationship between the activated carbon doping concentration (C_d) and resistivity (ρ) of the specimen is shown in Figure 11. The specimen's resistivity and doping concentration are in inverse proportion. At a temperature of 300 K, when the activated carbon doping concentration reaches 35 M, the electric conductivity increases rapidly, but when the activated carbon doping concentration increases to 45 M, the electric conductivity enters into a steady state. In the meanwhile, if the activated carbon doping concentration reaches 55 M, the adsorption of activated carbon is so strong that it causes the specimen's elasticity to reduce. Moreover, when the activated carbon doping concentration is 60 M, at which point adsorptive saturation is reached, many cracks are left on the surface of the hardened specimen. Increasing to 65 M causes adsorption over-saturation, and the activated carbon desorbs from the surface of the activated carbon doping concentration is derived as follows:

$$\rho = \begin{cases} (\rho_c - \rho_s) \frac{C_s - C_c}{C_d - C_c}, \text{ if } C_d > C_c \\ (\rho_o - \rho_c) \frac{C_c - C_d}{C_c - C_o}, C_d < C_c \end{cases}$$
(2)

In Formula (2), C_0 is the minimum doping concentration required for the elastic polymer composite materials to become resistance materials; ρ_0 is the resistivity of C_0 ; C_c is the critical doping concentration at which the conductor is in a steady state; ρ_c is the resistivity of the critical doping concentration; C_s is the adsorption-saturated doping concentration; and ρ_s is the resistivity of the adsorption-saturated saturated doping concentration. At a temperature of 300 K, for the elastic polymer composite materials doped with activated carbon, C_0 = 35 M, C_c = 45 M, C_s = 60 M, ρ_0 = 4.456 \times 10⁸ Ω -cm, ρ_c = 4 \times 10⁶ Ω -cm, and ρ_s = 1.12 \times 10⁴ Ω -cm.



Figure 11. The relationship between activated carbon doping concentration (C_d) and resistivity (ρ).

3.2. Graphite's Characteristic Equation

Figure 12 shows the relationship between the graphite doping concentration (C_d) and the resistivity (ρ) of the specimens. When the graphite doping concentration reaches 75 M, the electric conductivity increases sharply; when the graphite doping concentration reaches 79 M, the electric conductivity enters into a steady state; when the graphite doping concentration increases to 90 M, graphite reaches adsorptive saturation and the polymer's elasticity starts to deteriorate, and as a result, the specimen's edge has small cracks.



Figure 12. The relationship between graphite doping concentration and resistivity.

After many experiments, the equation for the resistance characteristic of the graphite doping concentration is derived as follows:

$$\rho = \rho_{\rm s} + \rho_0 \cdot e^{-Q_m \frac{C_d - C_o}{C_c - C_o}} \tag{3}$$

In Formula (3), C_0 is the minimum doping concentration required for the elastic polymer composite materials to become resistance materials; ρ_0 is the resistivity of C_0 ; C_c is the critical doping concentration at which the conductor is in a steady state; C_s is the adsorption-saturated doping concentration; ρ_s is the resistivity of the adsorption-saturated doping concentration; and Q_m is the quality factor of the material which is dimensionless, where the Q_m value of graphite is 4.

At a temperature of 300K, for the elastic polymer composite materials doped with graphite, $C_0 = 75$ M, $C_c = 80$ M, $C_S = 90$ M, $\rho_0 = 2.08 \times 10^6 \Omega$ -cm, and $\rho_s = 3.04 \times 10^2 \Omega$ -cm. When C_d is smaller than C_c , Equation (3) could be simplified as follows:

$$\rho = \rho_0 \cdot e^{-Q \frac{C_d - C_o}{C_c - C_o}} \tag{4}$$

4. Discussion

4.1. Temperature Coefficient of Resistance for Graphite's Physisorption

Since physisorption occurs without a chemical reaction, it does not need activation energy and can therefore occur easily and at low temperature. The adsorption is an exothermic reaction, so the resistance produced by the elastic polymer composite materials' physisorption is directly proportional to the temperature. The main force of physisorption is based on the weak van der Waals and easily affected by temperature, so temperature has a great impact on resistance. Conservation of energy dictates that electron transfer between an electrode and a redox system in solution or adsorbed on the electrode surface is fastest when the energy of the electron is equal in the metal and in the thermally activated redox system [43,44]. The elastic polymer composite materials were doped with a 46.3 M concentration of graphite, for which the influence of temperature on resistance is shown in Figure 13. The characteristic equations of resistivity (ρ) and temperature are as follows:

$$\rho = \begin{cases}
\rho_{0}, \text{ if } T_{e} < T_{0} \\
\rho_{0} + (\rho_{l} - \rho_{0}) \cdot e^{Q_{Td}D_{m}} \frac{(T_{e} - T_{0})}{T_{l} - T_{0}}, \text{ if } T_{e} < T_{l} \\
\rho = \left(\rho_{c} - \rho_{pl}\right) \cdot \frac{T_{c} - T_{e}}{T_{c} - T_{l}}, \text{ if } T_{pl} < T_{e} < T_{c} \\
(\rho_{nl} - \rho_{c}) \cdot \frac{T_{nl} - T_{c}}{T_{nl} - T_{c}}, \text{ if } T_{c} < T_{e} < T_{nl} \\
\rho_{c} \cdot e^{-Q_{Td}} \frac{(T_{e} - T_{c})}{T_{m} - T_{c}}, \text{ if } T_{e} > T_{c}
\end{cases}$$
(5)

In Formula (5), T_0 is the temperature at which absorption starts; ρ_0 is the resistivity at temperature T_0 ; T_{pl} is the initial temperature of the positive temperature coefficient on the linear region; ρ_{pl} is the resistivity when the temperature is at T_{pl} ; T_c is the critical temperature of the positive and negative temperature coefficients of the elastic polymer composite materials; ρ_c is the resistivity when the temperature is at T_c ; T_{nl} is the cutoff temperature of the negative temperature coefficient on the linear region; ρ_{nl} is the resistivity when the temperature is at T_{nl} ; T_m is the melting temperature of the elastic polymer composite materials; ρ_m is the resistivity when the temperature is at T_m ; T_e is the temperature of the testing environment; and Q_{Td} is the thermodynamics characteristic of the material that is dimensionless, and graphite's Q_{Td} is 2. The D_m is the thermal desorption characteristic of the material that is also dimensionless, and graphite's D_m is 3.



Figure 13. Resistivity and temperature characteristics of the elastic polymer composite materials doped with 46.3 M concentration of graphite.

For the elastic polymer composite materials doped with 46.3 M concentration of graphite, $T_0 = 267 \text{ K}$, $T_{pl} = 312 \text{ K}$, $T_c = 319 \text{ K}$, $T_{nl} = 322.5 \text{ K}$, $T_m = 343 \text{ K}$, $\rho_0 = 1.296 \times 10^3 \Omega$ -cm, $\rho_{pl} = 4.88 \times 10^4 \Omega$ -cm, $\rho_c = 5.712 \times 10^4 \Omega$ -cm, $\rho_{nl} = 4.936 \times 10^4 \Omega$ -cm, and $\rho_m = 7.04 \times 10^3 \Omega$ -cm.

4.2. Temperature Coefficient of Resistance for Activated Carbon's Physisorption

The influence of temperature on the resistance of the elastic polymer composite materials doped with activated carbon is shown in Figure 14. The characteristic equation of resistivity and temperature also accords with Formula (5). For the elastic polymer composite materials doped with 45M concentration of activated carbon, $Q_{Td} = 1.3$, $D_m = 7.7$, $T_0 = 277$ K, $T_{pl} = 333$ K, $T_c = 335$ K, $T_{nl} = 337$ K, $T_m = 343$ K, $\rho_0 = 2.856 \times 10^6 \ \Omega$ -cm, $\rho_{pl} = 9.384 \times 10^6 \ \Omega$ -cm, $\rho_c = 9.912 \times 10^6 \ \Omega$ -cm, $\rho_{nl} = 7.696 \times 10^6 \ \Omega$ -cm, and $\rho_m = 2.088 \times 10^6 \ \Omega$ -cm.

Since ρ_m is much higher than ρ_0 , the conductive characteristic of the elastic polymer composite materials is produced by physisorption. When the environmental temperature T_e is higher than T_c , the electronic kinetic energy of the valence band increases to cause the energy gap to become smaller. Meanwhile, the resistance temperature characteristics of the elastic polymer composite materials and the semiconductor are the same. Therefore, the resistance temperature coefficient of elastic polymer composite materials has the dual characteristics of metallic conductor and semiconductor.

For the physisorption produced by the elastic polymer composite materials doped with graphite or activated carbon, the resistivity and temperature coefficient characteristics have the same equation, that is, they have the same physical characteristics. However, the surface area of activated carbon is bigger, so the values of the material's thermodynamics characteristic (Q_{Td}) and thermal desorption characteristic (D_m) are different, which makes the resistance temperature characteristic curve of the activated carbon sharper than graphite.



Figure 14. Resistivity and temperature characteristics of the elastic polymer composite materials doped with 45 M concentration of activated carbon.

4.3. Application of Elastic Polymer Composite Materials Adsorption of Carbon Powder

Carbon elastic polymer composite materials exhibit a broad range of conductivities due to the diverse types of carbon materials available, allowing them to be utilized in electronic devices. This study leverages the high-surface-area properties of carbon materials to fabricate electronic components from carbon powder. This technology holds promise for widespread use in circuit components and the development of new electronic technologies. Additionally, the extremely low inductance of carbon elastic polymer composites makes them highly suitable for high-frequency circuits [45,46].

Carbon black (CB) is a preferred filler in large-scale industrial processes, primarily due to its cost-effectiveness. When added to elastic polymer composite materials, CB can impart electrical conductivity [47]. These systems exhibit percolation-type conductive behavior [48–50], and the introduction of fillers also significantly impacts the mechanical properties, making the material much tougher [51,52]. By selecting different fillers and adjusting the required conductivity levels, unique materials can be designed for various applications, including electromagnetic shielding, thermal resistors, automotive boards, power cable shielding, chemical vapor sensors, and pipe applications [53,54].

Highly conductive elastic polymer composite materials can be utilized to create both flexible printed circuit boards and traditional printed circuit boards. As illustrated in Figure 15, the process involves dissolving the material in an organic solvent and then using screen printing to form the conductive circuit. This method eliminates the need for metal materials like copper and avoids waste liquid from corrosion, thus reducing costs and benefiting the environment. These composite materials are also heat-resistant and adaptable to extreme climates, making them suitable for a wide range of applications.





Figure 15. The flexible circuit boards in this study are made from conductive elastic polymer composite materials.

5. Conclusions

This study focuses on the research and development of highly conductive elastic polymer composite materials. These materials are created by doping electrically insulating elastic polymers with high concentrations of activated carbon or graphite particles, which are then dispersed and compounded to form elastic composite conductive polymers. Activated carbon and graphite, both composed of carbon atoms, interact with the rubbery polymeric material through physisorption, altering the molecular packing and reducing the distance between polymer molecules. This results in a high residual force within the polymer molecules.

The performance of organic electronics has seen significant advancements over the past twenty years. Carbon powder allows molecules to stack tightly, reducing the energy gap and imparting semiconductor-like properties to polymers. When combined with solutionprocessed polymeric materials, such as those used in radio-frequency technologies, this development holds substantial practical value for electronic applications.

However, the electrical conductivity of doped activated carbon and graphite differs significantly. The key distinction lies in how their doping concentration relates to resistivity: in activated carbon, the relationship is linear, while in graphite, it follows an exponential pattern. Consequently, the ratio of surface area to particle size in carbon powder influences its conductive properties. When the surface area is large relative to particle size, the doping concentration and resistivity show a linear relationship. Conversely, if the ratio is low, the relationship becomes exponential. This characteristic is worth further investigation.

Additionally, this study revealed that the physisorption of elastic polymer composite materials is governed solely by van der Waals forces. Unlike covalent bonds, van der Waals forces are weaker and non-selective. While these forces do not form chemical bonds, they do produce a distinct infrared absorption spectrum.

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