Effects of Mott–Schottky Frequency Selection and Other Controlling Factors on Flat-Band Potential and Band-Edge Position Determination of TiO₂

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Abstract: The use of titanium dioxide for tackling environmental pollution has attracted great research interest recently. The potential of a photocatalyst in removing contaminants depends mainly on its conduction and valence-band edges relative to the standard potential of reactive oxygen species. One of the methods used in determining these band-edge positions is via the Mott–Schottky analysis. Thus, the aim of this work was to investigate the influence of the Mott–Schottky frequency and different electrochemical conditions on flat-band potential values and band-edge positions of pure anatase or Degussa P25, calcined or uncalcined in a 0.2 M Na₂SO₄ solution. The results showed that the shift in the flat-band potential was not so frequency-dependent in the Mott–Schottky analysis, however, other reasons, such as immersion duration leading to thenardite Na₂SO₄ salt deposition on the surface, irradiation of sources and the change in the pH of the solution (in the range of 1.64–12.11) were also responsible for it. In general, both the calcined anatase and P25 had a less negative value of the flat-band potential compared to the uncalcined. On the other hand, the calcined anatase had a tendency to have a less negative value of the flat-band potential than the calcined P25. From this study, the frequency range for obtaining the flat-band potential within one standard deviation in the Mott–Schottky at a single-frequency analysis was found to be between 200 and 2000 Hz. The energy difference between the Fermi level and the conduction band edge for anatase and P25, either calcined or uncalcined, was 0.097–0.186 and 0.084–0.192 eV, respectively. On the other hand, the band-edge position of anatase or P25 tended to shift upwards when it was repeatedly used in the photoelectrochemical analysis.

Keywords: Mott–Schottky; flat band potential; titanium dioxide; irradiation sources; pH; frequency of analysis; band-edge positions

1. Introduction

In recent years, titanium dioxide (TiO₂) has become the most extensively used metal oxide semiconductor in the field of environmental cleaning [1–4] and clean energy production [5–10]. The reasons for that mainly lie on the fact that TiO₂ has inherently unique properties, such as nontoxicity and great stability, as well as having a good photoactivity. The photoactivity of a semiconductor depends mainly on two essential parameters: the band-edge positions and the energy band-gap (E₉) [6,7,11,12]. The band-edge positions consist of conduction band edge (E₉) and valence band edge (Eᵥ). However, E₉ and Eᵥ cannot be measured directly using electrochemical methods. The estimation of E₉ and Eᵥ can be obtained via the determination of the flat-band potential (V₉) from electrochemical experiments. There are several electrochemical methods for obtaining V₉ [13], however, the Mott–Schottky analysis has been the most used because it permits V₉ and Nₙ (charge carrier density) to be determined, and subsequently, the band-edge positions can be constructed.

The Mott–Schottky analysis involves the total interfacial capacitances of a metal oxide semiconductor as a function of applied potentials, with each applied potential being cou-
plied with a range of perturbation frequencies. Traditionally, electrochemical impedance spectroscopy, involving a range of perturbation frequencies, has been adopted to obtain the interfacial capacitances for each applied potential. This Mott–Schottky electrochemical impedance spectroscopy (MS-EIS) analysis has been well-investigated for its complexities and pitfalls [13,14]. The following parameters have been set up for MS-EIS: (a) the range of applied potentials should not fall in the potential region where proton insertion would occur [15,16], and (b) the range of perturbation frequencies is typically between 10^5 and 0.1 Hz [13]. In the MS-EIS analysis, the impedance spectrum obtained at each applied potential will have to be fitted with a suitable equivalent circuit model to extract the values of space-charge capacitance (C_{SC}) at the thin film interface and Helmholtz capacitance (C_H) at the electrolyte interface based on a proper understanding of the interfacial behaviour across the range of the applied potentials. This technique has been well-investigated, however, it is usually time-consuming as it involves searching for a suitable equivalent circuit model for fixing the impedance spectrum corresponding to each of the applied potentials.

Thus, the adoption of Mott–Schottky at a single-frequency (MS-SF) analysis will usually be considered by researchers when C_H is considered negligible with N_d being less than 10^{23} m^{-2} [13], especially for undoped or low-doped semiconductors [17–19]. To obtain a nearly constant value of V_{fb} of TiO_2 based on the MS-SF analysis, some researchers proposed that a suitable range of frequencies should be between 10 and 1000 Hz [20,21], while others proposed between 500 and 1500 Hz [17,18,22] or even a higher frequency region between 5 and 45 kHz [12,23]. Until now, no systematic studies have properly addressed this issue. On the other hand, it has also been reported that V_{fb} is irradiation intensity-dependent [24] as well as pH-dependent, with a negative linear relationship being erected between them that follows the Nernst Equation [25–28].

In view of the above, a systematic study on frequency-dependent capacitance by Mott–Schottky analysis was undertaken, in which different TiO_2 thin films (anatase and P25—uncalcined and calcined at 400 °C) immersed in a 0.2 M Na_2SO_4 solution were characterized from 10 to 6000 Hz to investigate the influence of frequency on the determination of V_{fb}. This work revealed that the obtained V_{fb} is not so frequency dependent in an MS-SF analysis, and the thenardite Na_2SO_4 salt deposition on the surface of the nano-compact TiO_2 thin film after a prolonged MS-SF analysis was reported. Then, a suitable frequency was selected to be coupled with the MS-SF analysis to study the influence of irradiation sources, immersion duration, and solution pHs on the V_{fb} of pure Anatase. A relationship between V_{fb} vs. time under different irradiation sources and V_{fb} vs. pH of pure anatase in the laboratory light was presented. The plots between ΔE vs. frequency, irradiation sources, and pH, together with the band-edge positions for photocatalytic activity prediction, were also analyzed.

Theory on Band Edges Construction from MS-SF Analysis

The Mott–Schottky equation is principally based on the measurement of the total capacitances (C_T), which is the sum of the space-charge capacitance (C_{SC}), chemical capacitance (C_{µ}), and Helmholtz capacitance (C_H) induced at the n-type metal-oxide-semiconductor/solution interface over a range of applied potentials (V).

$$\frac{1}{C_T^2} = \frac{2}{\varepsilon e \varepsilon_0 A^2 N_d} \left( V - V_{fb} - \frac{k_B T}{e} \right)$$  \hspace{1cm} (1)

where e is the electronic charge (1.60 × 10^{-19} C), ϵ the relative permittivity of the n-type metal-oxide-semiconductor thin-film (where ϵ = 55 for TiO_2 thin-film [29]), ϵ_0 the vacuum permittivity (8.85 × 10^{-14} Fcm^{-1}), A the exposed surface area (cm), k_B the Boltzmann constant (1.38 × 10^{-23} m^2kgs^{-2}K^{-1}), and T the temperature in Kelvins.

C_{µ} needs to be taken into consideration in the calculation of C_T when the n-type metal-oxide-semiconductor thin-film has a nano size of roughly between 1 and 10 nm, involving discrete energy (quantum effect) [30], or the system is polarized to a sufficiently negative potential than V_{fb} that involves the proton insertion into TiO_2 thin-film [15,16].
Otherwise, $C_T$ usually consists of $C_{SC}$ at the thin-film interface and $C_H$ at the electrolyte interface, with both capacitances being put in series with each other:

$$\frac{1}{C_T} = \frac{1}{C_{SC}} + \frac{1}{C_H} \quad (2)$$

It is theoretically proven that for an $N_d$ less than $10^{23}$ m$^{-2}$, $C_H$ can be considered negligible, especially for undoped or low-doped semiconductors (n-type) [13]. Thus, $C_T \approx C_{SC}$. For n-type metal oxide semiconductors, when $C_H$ becomes negligible, the Mott–Schottky equation is, thus, as below:

$$\frac{1}{C_{SC}^2} = \frac{2}{e\varepsilon\varepsilon_0 A^2 N_d} \left( V - V_{fb} - \frac{k_B T}{e} \right) \quad (3)$$

In this case, $C_{SC}$ can then be calculated from the imaginary part of the measured impedance ($Z_{im}$) obtained over a range of applied potentials based on the following relation:

$$C_{SC} = \frac{1}{2\pi f Z_{im}} \quad (4)$$

where $f$ is a frequency that is coupled to the applied potentials in the MS-SF analysis.

A Mott–Schottky plot is thus displayed as a graph of $\frac{1}{C_{SC}^2}$ versus $V$, as shown in Figure 1a that is taken from one of the samples used in this study. The positive linear slope reveals that the nano-compact TiO$_2$ thin-film is an n-type metal oxide semiconductor. Given that $\frac{1}{C_{SC}^2} = 0$, Equation (3) becomes $V - V_{fb} - \frac{k_B T}{e} = 0$. Thus,

$$V_{fb} = V - \frac{k_B T}{e} \quad (5)$$

and

$$N_d = \frac{2}{e\varepsilon\varepsilon_0 A^2 m} \quad (6)$$

where $m$ is the slope of the Mott–Schottky plot in Figure 1a. The energy difference between $E_F$ and $E_C$ ($\Delta E$ in eV) in Figure 1b can be obtained as below:

$$\Delta E = E_C - E_F = k_B T \ln \left( \frac{N_d}{N_v} \right) \quad (7)$$

with $k_B$ being expressed in $8.62 \times 10^{-5}$ eVK$^{-1}$, $T$ in K, and $N_v$ being the effective density of the states in the conduction band, which can be calculated as shown below:

$$N_v = 2 \left( \frac{2\pi m^* k_B}{h^2} \right)^{3/2} \quad (8)$$

with $m^*$, the effective mass of the electron, being 0.1412 $m_0$ for the TiO$_2$ thin-film, where $m_0$ is $9.11 \times 10^{-31}$ kg, $k_B$ ($1.38 \times 10^{-23}$ m$^2$kg$^{-1}$K$^{-1}$), and $h$ the Planck constant ($6.63 \times 10^{-34}$ kgm$^{-2}$s$^{-1}$).

By referring to the standard hydrogen electrode (SHE), $V_{fb}$ is assigned as Fermi Level ($E_F$) [11,31], as demonstrated in Figure 1b. The $\Delta E$ is usually below 0.3 eV [11,31,32], which is small compared to $E_g$, which has a value of 3.30 eV for anatase [33], 3.02 eV for rutile [34], and 3.20–3.25 eV for P25 [6,7]. Thus, $V_{fb}$ is sometimes taken directly as $E_C$ for the band edge determination [35], while $E_g$ can be obtained from UV-Vis spectroscopy.
2. Experiment

2.1. Sample Preparation

Two types of commercial TiO$_2$ were adopted, which were pure anatase (Hombikat UV-100, Sachtleben Chemie, Germany) or Degussa P25 (AEROXIDE® TiO$_2$, Evonik, Germany), which has 78% anatase and 14% rutile \cite{36}. About 2 g of anatase or P25 powder was mixed with 4 mL of distilled water and stirred using a glass rod for around 10 min to form a paste. The paste was then coated on a fluorine-doped tin oxide (FTO) glass slide (OSSILA TEC15: 20 × 15 × 2.2 mm, FTO thickness of 200 nm, 12.46 ± 0.50 Ω/sq) using the doctor blade technique to obtain FTO/TiO$_2$ samples. The FTO substrate was initially taped with two layers of cello tape to limit the thickness as well as the exposed surface area to 1 cm$^2$. The paste coated on the FTO was then calcined at 400 °C for one hour in the oven with the ramp of 20 °C/min, and this was marked as Ana(400) or P25(400), or it was left to dry in the air for 24 h at the room temperature, and this was marked as Ana or P25. The temperature of 400 °C will not change anatase crystallite to rutile. In addition, it will help to improve the electronic contact between all the TiO$_2$ particles as well as between TiO$_2$ and FTO \cite{37,38}, leading to a more stable adhesion to FTO \cite{11}. The stability of the calcined TiO$_2$ thin film at 400 °C was also tested by gently stirring the solution where the sample was immersed with a magnet. No sign of flaking off was observed after the two-hour stirring. The reuse of the calcined TiO$_2$ thin film in another test was guaranteed for its stability, especially in the solution-pH test carried out in this study, where re-immersing the sample was needed after the desired solution pH was achieved. For the TiO$_2$ thin film treated at room temperature, gentle immersion in the solution was required to guarantee its stability.

2.2. Experimental Setup and Electrochemical Method

A three-electrode electrochemical cell was used, where the working electrode was the corresponding sample of FTO/TiO$_2$, the reference electrode was Ag/AgCl (3 M KCl), and the counter electrode was a platinum wire. The solution was 0.2 M Na$_2$SO$_4$ at a pH of 6.5. Potentiostat/Galvanostat (PGSTAT204, Metrohm, Herisau, Switzerland) was adopted to perform the MS-SF analysis at the DC potential starting from 1 to −1 V with an AC amplitude of ±10 mV coupling with a frequency. All the MS-SF tests described below were carried out at a room temperature of 21 °C.

In order to solve the doubt of the selection of a suitable frequency for the $V_{fb}$ determination in MS-SF analysis, the frequency that would be coupled in MS-SF test was chosen from 10 (starting frequency) to 6000 Hz (ending frequency). In total, 15 frequencies were chosen between 10 and 6000 Hz. For this study, seven samples were made and labelled as in Table 1 and all tests were performed in the laboratory light. The $V_{fb}$ data were then analyzed to view which frequency range is suitable to be adopted in MS-SF analysis.

![Figure 1.](image-url)
Table 1. The sample’s label and its description in the determination of a suitable frequency used in the MS-SF analysis.

<table>
<thead>
<tr>
<th>Sample’s Label</th>
<th>Description</th>
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<tbody>
<tr>
<td>Ana(400)-fs&amp;fs</td>
<td>Anatase calcined at 400 °C, fresh sample, and fresh solution for each frequency tested</td>
</tr>
<tr>
<td>Ana(400)-ss&amp;ss</td>
<td>Anatase calcined at 400 °C, same sample, and same solution throughout the test</td>
</tr>
<tr>
<td>Ana-ss&amp;ss</td>
<td>Anatase in as-received, same sample, and same solution throughout the test</td>
</tr>
<tr>
<td>P25-fs&amp;fs</td>
<td>P25 calcined at 400 °C, fresh sample, and fresh solution for each frequency tested</td>
</tr>
<tr>
<td>P25(400)-fs&amp;fs</td>
<td>P25 calcined at 400 °C, fresh sample, and fresh solution for each frequency tested</td>
</tr>
<tr>
<td>P25(400)-ss&amp;ss</td>
<td>P25 calcined at 400 °C, same sample, and same solution throughout the test</td>
</tr>
<tr>
<td>P25(400)-ss&amp;ss + stir</td>
<td>P25 calcined at 400 °C, same sample, and same solution plus solution stirring by magnet throughout the test</td>
</tr>
</tbody>
</table>

After the suitable frequency range was determined, a frequency was then selected within the range to perform the MS-SF test on Ana(400) to study the influence of different irradiation sources (dark, laboratory light, and UV (Philips TL-D 15 W BLB: 47 mWcm$^{-2}$)), immersion duration, and solution pHs on $V_{fb}$. For the solution-pH test in the laboratory light, two separate tests were performed—one was to increase the pH using aqueous sodium hydroxide ($\text{(NaOH)}_{aq}$), and another was to decrease the pH using acid hydrochloric (HCl) from the original pH of 0.2 M Na$_2$SO$_4$ solution of 6.5. After each drop of ($\text{(NaOH)}_{aq}$ or HCl, the solution in the electrochemical cell was magnetically stirred. When the determined pH was reached, the same sample was immersed back in the solution, and the MS-SF test was performed.

After the MS-SF tests, the samples were left dry in the air and analyzed ex situ by X-ray diffraction (D8 Advance, Bruker, Billerica, MA, USA) and scanning electron microscopy with energy dispersive spectroscopy (S-4800, Hitachi, Tokyo, Japan). A UV-Vis spectrometer (UV-2600, Shimadzu) was used to obtain the band gap energies ($E_g$) of Ana(400), Ana, P25(400), and P25 powder that would be used to construct the band-edge positions of TiO$_2$.

3. Results and Discussion

3.1. Sample Characterization

Figure 2 shows the thermogravimetric analysis (TGA) of as-received anatase and P25. When the powders are heated from 0 to 450 °C, a weight loss of about 13.32 and 2.57% was observed for anatase and P25, respectively. The weight loss is mainly due to the elimination of the residual water and organic materials in both powders [39]. No weight loss was found from 450 to 1000 °C, indicating that both powders have high thermal stability. Thus, the sample prepared by calcining at 400 °C for one hour in this study would not alter its microstructural property.

The powders of Ana, Ana(400), P25, and P25(400) were analyzed in the dry state using a laser diffractometer for their particle size distribution, as shown in Figure 3a. It can be seen that in the dry state, the particle sizes range from 0.1 to 100 µm, with both Ana and Ana(400) having the same maximum peaks at 1.2 µm; however, P25 and P25(400) had the maximum peaks at 2.6 and 2.9 µm, respectively. This indicates that the calcination at 400 °C does not shift the range of particle sizes much. For the photoelectrochemical analysis in this study, the powder was first mixed with distilled water, which usually leads to agglomeration, and thus, the magnetic stirring of the mixture is necessary to separate the powders in the paste. The microstructure and the thickness of the paste after being coated on the FTO using the doctor blade technique were examined under SEM. Figure 3b shows one example of the side morphology of the paste coated on FTO, which is P25. The average thickness of P25 is found to be 40.6 ± 1.2; P25(400), 35.9 ± 1.6; Ana, 37.1 ± 3.6; and Ana(400), 33.5 ± 4.8 µm. Figure 3c–f show the top morphology of the nano-compact P25, P25(400), Ana, and Ana(400), respectively, with the particle sizes being found to be between 51.6 and 82.4 nm for P25, 43.7 to 55.2 nm for P25(400), 41.7 to 78.4 nm for Ana, and 32.3 to 70.4 nm for Ana(400). All the particle sizes are bigger than 10 nm, wherever the discrete energy (quantum effect) can be neglected [30].
Figure 2. Thermogravimetric analysis of as-received anatase and P25.

Figure 3. (a) Particle size distribution measured by laser diffraction. (b) Side morphology revealing the thickness of nano-compact-P25 thin film. Top morphology revealing particle sizes of nano-compact thin films of (c) P25, (d) P25(400), (e) Ana, and (f) Ana (400).
3.2. Frequency Determination on MS-SF Analysis

$V_{fb}$ of the different samples (Table 1) are plotted against the frequency coupled in the MS-SF analysis, as shown in Figure 4. For Ana(400)-fs&fs, P25-fs&fs, and P25(400)-fs&fs, either uncalcined or calcined, $V_{fb}$ does not shift tremendously towards negative values when higher frequencies are adopted, showing no clear trends of frequency dependence. However, for Ana(400)-ss&ss, Ana-ss&ss, P25(400)-ss&ss, and P25(400)-ss&ss + stir, either uncalcined or calcined, with the same sample and same solution being applied throughout the test, $V_{fb}$ shows an obvious steep shift to a more negative value with the increase in frequency. Concerning the effect of calcination, a comparison between Ana(400)-ss&ss and Ana-ss&ss as well as P25(400)-fs&fs and P25-fs&fs is made, and it can be seen that the calcined samples at 400 °C tend to have a less negative $V_{fb}$ than those that are uncalcined, regardless of whether they are fs&fs or ss&ss.

Figure 4. Flat-band potential ($V_{fb}$) as a function of a frequency range from 10 to 6000 Hz for nano-compact Ana, Ana(400), P25, and P25(400) thin film in 0.2 M Na$_2$SO$_4$ at pH 6.5 in the laboratory light.

The samples, after MS-SF analysis, were examined under XRD and SEM-EDX. The XRD patterns obtained for all samples show the presence of Na$_2$SO$_4$ peaks on the surface of the thin film [40,41]. Since the film is thin and has cracks due to the drying process, other peaks belonging to FTO, which is SnO$_2$ [42], are also detected. In general, the obtained XRD patterns of the studied samples can be divided into two groups: (i) peaks displaying the presence of anatase, Na$_2$SO$_4$, and SnO$_2$ and (ii) peaks displaying the presence of anatase, rutile, Na$_2$SO$_4$, and SnO$_2$. Thus, only two XRD patterns are shown here as a representative for each group, with Figure 5a being Ana(400)-ss&ss, which exemplifies a clearly crystalline phase of anatase, Na$_2$SO$_4$, and SnO$_2$; Figure 5b being P25(400)-ss&ss + stir that exemplifies a clearly crystalline phase of anatase-rutile [43], Na$_2$SO$_4$, and SnO$_2$. 
Here, only the SEM image, mapping, and EDX spectrum of P25(400)-ss&ss + stir, as displayed in Figure 6, is adopted as a representative of all samples because the same salt was found deposited on their surface after the MS-SF test. The SEM image on the right side of Figure 6a is the magnified microstructure of Na$_2$SO$_4$ at 5000×. The elemental mappings of Ti in pink, O in green, Na in purple, and S in light blue in Figure 6b clearly confirm that the salt is thenardite Na$_2$SO$_4$, which is deposited through heterogeneous nucleation on the surface of the TiO$_2$ thin film [44]. Figure 6c shows the full EDS spectrum of the SEM image on the right side of Figure 6a. The full EDS spectrum also reveals the same elements of Ti, O, Na, and S.
Figure 6. (a) SEM images of the nano-compact P25(400)-ss&ss + stir thin film. (b) The EDS mappings of the elements of Ti (pink), O (green), Na (purple), and S (blue). (c) The full EDS spectrum of the right SEM image of (a) at ×5000.

The intensity at [111] peak of thenardite Na$_2$SO$_4$ in the XRD pattern at $2\theta$ of 19.0° is plotted in Figure 7 as a rough estimation of the differences in the amount of thenardite Na$_2$SO$_4$ salt deposited on the surface, without any attempts of quantification. In general, samples under the condition of ss&ss present a higher intensity than fs&fs. Figure 4 shows that the $V_{fb}$ of samples with the ss&ss condition tends to have a steep drop to more negative values compared to fs&fs from 10 to 6000 Hz. Thus, the deposition of the thenardite Na$_2$SO$_4$ salt could be responsible for the shifting of the $V_{fb}$ to a more negative value.

Figure 7. Intensity at XRD peak [111] of thenardite Na$_2$SO$_4$ to estimate the amount of thenardite Na$_2$SO$_4$ salt deposited on the TiO$_2$ surface.

In order to determine the suitable frequency to be adopted in the MS-SF analysis, the mean of $V_{fb}$ for seven samples, studied in Figure 4, over a range of frequencies from 10 to 6000 Hz and their standard deviation, $\sigma$, were calculated. The range of frequency corresponding to the mean $V_{fb} \pm 1\sigma$ is displayed in Table 2. By performing a study on the overlapping of the frequency range for these seven cases, the suitable frequency range for obtaining $V_{fb} \pm 1\sigma$ using an MS-SF analysis is found to be between 200 and 2000 Hz.
3.3. Influence of Different Irradiation Sources

Different irradiation sources, which are UV, lab light, or dark, were adopted and scanned from the potential of 1 to −1 V, with each scan lasting 12 min, using a frequency of 400 Hz. The same scan was repeated eight times on the Ana(400)-ss&ss thin film to study the effect of immersion duration. It can be seen that different irradiation sources pose different values on the \( V_{fb} \) of Ana(400)-ss&ss. An obvious deep shift of \( V_{fb} \) to a negative value is observed with time under the UV, laboratory light, or dark, as shown in Figure 8a. At the eighth scan, the differences in the \( V_{fb} \) of UV, laboratory light, and dark are bigger than the differences in \( V_{fb} \) at the first scan. When the sample is examined under XRD, it reveals that the intensity at peak [111] of thenardite is higher when the irradiation source is UV, followed by laboratory light and the dark, as shown in Figure 8b. Figure 8c–e are SEM microstructures, SEM mappings, and an EDX spectrum showing the thenardite Na\(_2\)SO\(_4\) salt on the surface of Ana(400)-ss&ss under the UV, respectively. The samples under the laboratory light or in the dark have similar characteristics to those under UV. Thus, the shift of \( V_{fb} \) to a negative value could be, again, due to the deposition of a layer of thenardite Na\(_2\)SO\(_4\) salt, instead of the influence of the application of different types of irradiation sources.

![Figure 8. Cont.](image-url)
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3.4. Influence of pH

The effect of the pH on the $V_{fb}$ of Ana (400)-ss&ss is given in Figure 9. It has also been reported that the value of $V_{fb}$ is pH-dependent, with a negative linear relationship being erected between them based on the Nernst Equation [25,27,28]. However, in this study, when the pH of 0.2 M Na$_2$SO$_4$ solution is increased or decreased from its starting pH of 6.5 using (NaOH)$_{aq}$ or HCl, respectively, the obtained graph displays three distinct slopes that can be indicated by A, B, and C, which is contrary to the reported linear Nernstian relationship between $V_{fb}$ and pH [25,28]. However, the values of $V_{fb}$ in the strong acidic region and strong basic region, which are $-0.19$ at pH 1.64 and $-0.97$ V at pH 12.11, respectively, are indeed consistent with others [23,45]. Although TiO$_2$ is inert to the pH change, its surface tends to adsorb hydrogen ions (H$^+$) or hydroxide ions (OH$^-$) in the function of the pH. Thus, the less negative values of $V_{fb}$ in the acidic solution are attributed to the adsorption of H$^+$ on the surface O atoms leading to the formation of a positively charged layer, and the more negative values of $V_{fb}$ in the basic solution are caused by the adsorption of OH$^-$ on the surface Ti atoms forming a negatively charged layer [46].
From the pH of 1.6 to 4.5, \( V_{fb} \) shows a steep decrease with a negative slope of 0.1282 V/pH. From the pH of 9.6 to 12.1, \( V_{fb} \) shows a steep decrease as well, with a negative slope of 0.1371 V/pH. While in the mildly acidic, neutral, and mildly basic region, which is from pH 4.5 to 9.6, \( V_{fb} \) shows almost a plateau somewhere between \(-0.5\) to \(-0.6\) V [12], with a negative slope of 0.0252 V/pH. The pH of zero change (pH\text{pzc}) for the nanoparticle of TiO\(_2\) is reported to be from 4.4–5.4 [47], 6.2 [48], 6.5 [49], and 7.0 [50], and this could explain the plateau formed between pH 4.5 and 9.6. Gerischer [27] reported that the theoretical value of the slope should be 0.059 V/pH, given the fact that both charged species need to appear on the surface in order to fulfill the Nernstian relationship. Thus, this could possibly be the reason that the slope of A (strong acidic region) and the slope of C (strong basic region) are far from 0.059 V/pH. The experimentally reported value of the slope fulfilling the Nernstian relationship is somewhere between \(-0.056\) and \(-0.065\) V/pH [27,28]. Applying the linear regression to the whole data points (indicated as D) gives the slope of 0.065 V/pH [27,28]. Applying the linear regression to the whole data points (indicated as D) gives the slope of 0.065 V/pH [27,28].

### 3.5. Band Gap

The Kubelka–Munk absorption coefficient (\( F(R_{\infty}) \)) of the sample obtained from the UV-Vis spectrometer is linked to the incident photon energy (\( h\nu \)) and band gap energy (\( E_g \)) using the Tauc equation:

\[
(F(R_{\infty}) \cdot h\nu)^{1/\gamma} = h\nu - E_g, \tag{9}
\]

where \( h \) is the Planck constant, \( \nu \) the photon’s frequency, and the factor \( \gamma \) relates to the types of electron transition. By taking \( \gamma = 2 \), which is an indirect transition band gap, (\( F(R_{\infty}) \cdot h\nu \))\(^{1/2} \) versus \( h\nu \) is plotted, as shown in Figure 10. The \( E_g \) of Ana(400), Ana, P25(400), and P25 powder are then obtained from the \( x \)-axis intersection point of the plot, which are 3.30, 3.33, 3.21, 3.22 eV, respectively, and are in agreement with the reported Literature [6,7,33].
around 0.124 to 0.128.

ever, in the region of strong acid 

Figure 10. Plot of \((F(R_{∞})\cdot h\nu)^{1/2}\) and \(h\nu\) for obtaining band gap energies (\(E_g\)) of Ana(400), Ana, P25(400), and P25 powder.

### 3.6. Band-Edge positions for Photocatalytic Activity Prediction

The band-edge positions of all the studied samples were constructed by first converting the \(V_{fb}\) (reference electrode Ag/AgCl used in this study) to the \(V_{fb}\) in SHE, which is assigned as \(E_F\) [11], as shown in the schematic diagram in Figure 1.

The \(N_d\) obtained from the Mott–Schottky plots based on Equation (6) is between the order of 19 and 21 (Tables S1–S11), fulfilling the requirement that \(C_H\) can be considered negligible in the calculation of \(C_F\) [13].

The \(\Delta E\) (Tables S1–S11) was calculated from \(N_d\) obtained from the MS-SF plots using Equations (7) and (8).

The plots of \(\Delta E\) versus frequency, irradiation sources, immersion duration, and solution pH are shown in Figure 11. Figure 11a shows that uncalcined TiO\(_2\) (Ana-ss&ss and P25-fs&fs) tends to have almost a constant value of \(\Delta E\) that is not greatly influenced by the frequency adopted in the MS-SF analysis and is basically lower than that of the calcined at 400 °C. However, the \(\Delta E\) of the calcined TiO\(_2\) tends to show a decrease with the increase in the frequency adopted, regardless of whether it is a fresh or the same sample throughout MS-SF analysis. Ana(400)-fs&fs has \(\Delta E\) between 0.097 and 0.186; Ana(400)-ss&ss, 0.120–0.175; Ana-ss&ss, 0.094–0.104; P25-fs&fs, 0.084–0.108; P25(400)-fs&fs, 0.114–0.191; P25(400)-ss&ss, 0.118–0.192; and P25(400)-ss&ss + stir, 0.122–0.182. Figure 11b shows that the \(\Delta E\) is larger in the dark (0.152–0.170 eV), followed by the UV (0.139–0.144 eV) and the laboratory light (0.109–0.114 eV). In the dark, \(E_F\) shifts close to the valence band, and thus, this explains how \(\Delta E\) is larger in the dark. Supposedly, TiO\(_2\) under the UV should have a smaller \(\Delta E\) when compared to that of laboratory light. However, this work reveals that \(\Delta E\) under the intense UV has a larger value than in the laboratory light which contains a lesser intensity of UV. This could be due to the accumulation of the excited electrons in the conduction band leading to a Burstein shift [35], causing the \(\Delta E\) of UV to become larger than that of the laboratory light. For the pH influence, as shown in Figure 11c, it can be seen that \(\Delta E\) in the mildly acidic region is larger than that in the mildly basic region. However, in the region of strong acidity and strong basicity, the value of \(\Delta E\) becomes nearly the same, around 0.124 to 0.128.
By adding \( \Delta E \) to \( E_F \), the \( E_C \) of nano-compact TiO\(_2\) thin films in different electrochemical conditions were obtained. With the band gaps from Figure 10, each corresponding \( E_V \) was then calculated (see Tables S1–S11). The band-edge positions of Ana(400), Ana, P25(400), and P25 thin films obtained under different electrochemical conditions using MS-SF analysis are then displayed in Figure 12, with the pH-indication of the standard potentials of chemical reactions corresponding to water splitting and reactive oxygen species formation.

In Figure 12a, samples are tested at pH 6.5 under laboratory light and divided into two groups, which include ss&ss (under the condition of the same sample and same solution throughout the test) and fs&fs (under the condition of fresh sample and the fresh solution being applied for each frequency tested). It can be seen that Ana(400)-ss&ss, Ana-ss&ss, and P25(400)-ss&ss tend to show an obvious upward shift in their band-edge positions from 10 to 6000 Hz, while Ana(400)-fs&fs, P25-fs&fs, and P25(400)-fs&fs tend to show a slight upward shift. Despite that, they do not lose their photocatalytic ability in hydrogen and oxygen production via water splitting at pH 6.5 because their \( E_C \) is well above the corresponding pH-indicated \( H_2O/H_2 \) potential, and the \( E_V \) is well below the corresponding pH-indicated \( H_2O/O_2 \) potential [16,51,52].

**Figure 11.** (a) \( \Delta E \) of nano-compact Ana(400)-fs&fs, Ana(400)-ss&ss, Ana-ss&ss, P25-fs&fs, P25(400)-fs&fs, P25(400)-ss&ss, and P25(400)-ss&ss + stir thin film coated on FTO, as a function of a frequency range from 10 to 6000 Hz in the laboratory light. (b) \( \Delta E \) of nano-compact Ana(400)-ss&ss thin film under different irradiation sources. (c) \( \Delta E \) of nano-compact Ana(400)-ss&ss thin film under different pH solutions in the laboratory light.
For Ana(400)-ss&ss, under UV at pH 6.5, very little upward shift is observed after the 96 min immersion in Figure 12b; however, under the laboratory light or in the dark, an obvious upward shift in the band-edge positions is clearly seen when the immersion duration becomes longer. Although this is the case, there is no loss in their photocatalytic ability in hydrogen and oxygen production via water splitting. As for the influence of the solution pH, the band-edge positions in the strongly basic regions are found to be higher than those in the strongly acidic regions. From pH 4.46 to 9.64, the band edges have similar positions. However, in the strongly acidic solution below pH 2.17, TiO$_2$ can fail to produce H$_2$ at $E_C$ because it locates slightly below the corresponding pH-indicated H$^+$/H$_2$ potential.

The superoxide radical anion (O$_2^-$), with its reported standard reduction potential ranging from $-0.27$ to $-0.33$ V vs. SHE in alkaline solution [53,54], is reportedly effective in killing microorganisms [55]. It is mainly produced from the reduction of dissolved oxygen gas at the $E_C$ of TiO$_2$ with the photogenerated electron ($O_{2(aq)}^- + e^- \leftrightarrow O_{2(aq)}^-$). Further reduction of O$_2^-$ with the photogenerated electron will lead to the formation of hydrogen peroxide ($O_{2}^-(aq) + 2H^+ + 2e^- \rightarrow H_2O_2$). The H$_2$O$_2$ then undergoes one-electron
reduction and forms a hydroxyl radical \((H_2O_2 + H^+ + e^- \rightarrow H_2O + OH^-)\), which is the most reactive species that oxidize most of the inorganic and organic pollutants \([56,57]\). On the other hand, the photogenerated hole \((h^+)\) at the \(E_V\) of \(TiO_2\) can oxidize the water molecules to form \(OH^+\) \((H_2O_{(aq)} + h^+ \rightarrow OH^+_{(aq)} + H^+)\) \([53,58]\), with its reported standard oxidation potential being \(2.80\ V\) in the acidic solution \((pH\ 0)\) \([52]\), \(1.86\ V\) in the basic solution \((pH\ 14)\) \([52]\), and \(2.27\ V\) in the neutral solution \((pH\ 7)\) \([53]\).

Through the comparison of the samples of \(ss&ss\) and \(fs&fs\) at \(pH\ 6.5\), it can be seen that the application of frequency from low to high, in Figure 12a, obviously increases the band-edge positions of \(ss&ss\). The longer immersion in the laboratory light or in the dark, in Figure 12b, also leads to an obvious increase in the band-edge positions of \(ss&ss\). However, \(ss&ss\) does not lose its photocatalytic ability in superoxide radical anion and hydroxyl radical production. This is because all \(Ev\) are well below the corresponding pH-indicated \(H_2O/OH^-\) potential, and the \(E_C\) is well above the corresponding pH-indicated \(O_2/O_2^-\) potential. In Figure 12b, the band-edge positions of \(Ana(400)-ss&ss\) are higher in strongly basic solutions than in strongly acidic solutions; however, it shows a plateau from \(pH\ 4.46\) to \(9.64\). In strongly acidic solutions below the \(pH\ 2.97\), \(Ana(400)-ss&ss\) loses its photocatalytic ability in \(O_2^-\) generation because \(E_C\) is well below the pH-indicated \(O_2/O_2^-\) potential. This can be because, in acidic solutions, the dissolved \(O_2\) is prone to react with the photogenerated electron, together with the existing proton in the solution, to form a perhydroxyl radical \((HO_2^-)\) \([54]\).

4. Conclusions

This preliminary study on the effects of frequency selection in Mott–Schottky analysis on the flat-band potential determination revealed the precipitation of thernardite \(Na_2SO_4\) salt from a \(0.2\ M\ Na_2SO_4\) solution on the surface of the nano-compact \(TiO_2\) thin film after the test, and this could cause a negative shift in the flat-band potential more than its frequency dependence, as previously reported. This study on the overlapping of frequency ranges for the analyzed cases showed that the suitable frequency range for obtaining \(V_{fb} \pm 1\sigma\) using Mott Schottky at a single-frequency analysis was found to be between \(200\) and \(2000\ Hz\). On the other hand, irradiation sources, immersion duration, and solution pHs contributed as well to the shift, with the pH of the solution being the most influential parameter on the flat-band potential and the band-edge positions of the nano-compact \(TiO_2\) thin film immersed in the solution. According to the analysis of band-edge positions, the nano-compact \(TiO_2\) thin film would lose its photocatalytic ability in superoxide radical anion generation at a \(pH\) below \(2.97\), and it would fail to produce \(H_2\) at a \(pH\) below \(2.17\).

Supplementary Materials: The following supporting information can be downloaded at: https://www.mdpi.com/article/10.3390/catal13061000/s1, Table S1: \(Ana(400)-fs&fs\); Table S2: \(Ana(400)-ss&ss\); Table S3: \(Ana-ss&ss\); Table S4: \(P25-fs&fs\); Table S5: \(P25(400)-fs&fs\); Table S6: \(P25(400)-ss&ss\); Table S7: \(P25(400)-ss&ss+stir\); Table S8: \(Ana(400)-ss&ss\) under UV; Table S9: \(Ana(400)-ss&ss\) under laboratory light; Table S10: \(Ana(400)-ss&ss\) under the dark; Table S11: \(Ana(400)-ss&ss\) at Different pHs.

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