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Abstract: This paper presents the effect of a composite poly(3,4-ethylenedioxythiophene) polystyrene sulfonate PEDOT:PSS and copper-doped nickel oxide (Cu:NiO<sub>x</sub>) hole transport layer (HTL) on the performance of perovskite solar cells (PSCs). Thin films of Cu:NiO<sub>x</sub> were spin-coated onto fluorinedoped tin oxide (FTO) glass substrates using a blend of nickel acetate tetrahydrate, 2-methoxyethanol and monoethanolamine (MEA) and copper acetate monohydrate. The prepared solution was stirred at 65 °C for 4 h and spin-coated onto the FTO substrates at 3000 rpm for 30 s in a nitrogen glovebox. The Cu:NiOx/FTO/glass structure was then annealed in air at 400 °C for 30 min. A mixture of PEDOT:PSS and isopropyl alcohol (IPA) (in 1:0.05 wt%) was spun onto the Cu:NiOx/FTO/glass substrate at 4000 rpm for 60 s. The multilayer structure was annealed at 130 °C for 15 min. Subsequently, the perovskite precursor (0.95 M) of methylammonium iodide (MAI) to lead acetate trihydrate (Pb(OAc)<sub>2</sub>·3H<sub>2</sub>O) was spin-coated at 4000 rpm for 200 s and thermally annealed at 80 °C for 12 min. The inverted planar perovskite solar cells were then fabricated by the deposition of a photoactive layer (CH3NH3PbI3), [6,6]-phenyl C61-butyric acid methyl ester (PCBM), and a Ag electrode. The mechanical behavior of the device during the fabrication of the Cu:NiO<sub>x</sub> HTL was modeled with finite element simulations using Abaqus/Complete Abaqus Environment CAE. The results show that incorporating Cu:NiOx into the PSC device improves its density-voltage (J-V) behavior, giving an enhanced photoconversion efficiency (PCE) of ~12.8% from ~9.8% and ~11.5% when PEDOT:PSS-only and Cu:NiOx-only are fabricated, respectively. The short circuit current density  $J_{sc}$  for the 0.1 M Cu:NiO<sub>x</sub> and 0.2 M Cu:NiO<sub>x</sub>-based devices increased by 18% and 9%, respectively, due to the increase in the electrical conductivity of the Cu:NiO<sub>x</sub> which provides room for more charges to be extracted out of the absorber layer. The increases in the PCEs were due to the copper-doped nickel oxide blend with the PEDOT:PSS which enhanced the exciton density and charge transport efficiency leading to higher electrical conductivity. The results indicate that the devices with the copper-doped nickel oxide hole transport layer (HTL) are slower to degrade compared with the PEDOT:PSS-only-based HTL. The finite element analyses show that the Cu:NiO<sub>x</sub> layer would not extensively deform the device, leading to improved stability and enhanced performance. The implications of the results are discussed for the design of low-temperature solution-processed PSCs with copper-doped nickel oxide composite HTLs.

**Keywords:** hybrid HTL; nickel oxide; copper-doped nickel oxide; perovskite solar cells; efficiency; finite element simulation



Citation: Asare, J.; Sanni, D.M.; Agyei-Tuffour, B.; Agede, E.; Oyewole, O.K.; Yerramilli, A.S.; Doumon, N.Y. A Hybrid Hole Transport Layer for Perovskite-Based Solar Cells. *Energies* **2021**, *14*, 1949. https://doi.org/10.3390/en14071949

Academic Editors: Senthilarasu Sundaram and Jaemin Kong

Received: 27 January 2021 Accepted: 22 March 2021 Published: 1 April 2021

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## 1. Introduction

Organic–inorganic halide perovskite materials have attracted significant attention because of their excellent photoelectrical performance [1]. These materials were initially used in 2009 as semiconductor sensitizers in dye-sensitized solar cells [2–5], with a material configuration of MAPbX<sub>3</sub>, where M is the methyl (organic part), A is the ammonia (inorganic part), and X could be Br<sup>-</sup>, Cl<sup>-</sup> or I<sup>-</sup>. The merits of these materials include solution-phase processing, intense broad-band absorption, low exciton binding energy and long charge-diffusion length [6–8]. These advantages have enabled perovskite materials to be rapidly developed as a new photovoltaic (PV) technology. Such perovskite materials are the most competitive absorbers for highly efficient solar cell devices [6–11]. The perovskite solar cell (PSC) has reached a power conversion efficiency (PCE) of 25.2% [12] due to its tunable bandgap, absorption coefficient, and ability to form thin films with excellent charge-carrier transport properties with an ostensive tolerance to defects [7,13–15]. Recent development in perovskite and silicon tandem solar cells has revealed efficiencies of 29.15% [16]. These features make such PSCs commercially promising [17–19].

Several techniques have been used to deposit perovskite films. These include the vapor deposition techniques [8,20–24]. The single-step spin and two-step sequential deposition methods are both solution-processed, cost-effective production methods. Perovskite films can be obtained from different lead sources such as lead iodide (PbI<sub>2</sub>) and lead acetate (Pb( $C_2H_3O_2)_2$ ). Recently, perovskite solar devices have been produced with lead acetate as the main source of lead [25,26]. Lead acetate has also attracted attention as a source material for perovskite solar cells because it speeds up the crystal growth of perovskite films due to the facile removal of *N*-methylammonium acetate (CH<sub>3</sub>NH<sub>3</sub>Ac) [27]. As a result, smooth films with fewer pinholes are obtained, which results in improved device performance.

This paper explores poly(3,4-ethylenedioxythiophene) polystyrene sulfonate-nickel oxide PEDOT:PSS/Cu:NiO<sub>x</sub>-based composites as an alternative exciton generation and transport path to improve electron blocking capability and performance compared to purely PEDOT:PSS hole injection-based devices [28]. Light passing through the transparent electrodes onto the photosensitive perovskite material (CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub>) stimulates excitations called electron–hole pairs. These charged particles, when separated, will diffuse through the PEDOT:PSS/Cu:NiO<sub>x</sub> composite and the [6,6]-phenyl C<sub>61</sub>-butyric acid methyl ester (PCBM) layers to their respective electrodes to generate an electric current [29]. The schematic view of the device structure is shown in Figure 1a. Note that the control devices are made of PEDOT:PSS-only and copper-doped nickel oxide (Cu:NiO<sub>x</sub>)-only. The PEDOT:PSS/Cu:NiOx composite hole transport layer is shown in Figure 1b. The device that demonstrated the best functionality was achieved with the 0.1 M Cu:NiO<sub>x</sub> composite hole transport layer (HTL).

Nickel Oxide NiO is an inorganic p-type material commonly used in semiconductor fabrication. It has a large bandgap and possesses very deep valence bands. These align perfectly with the highest occupied molecular orbital HOMO levels of a variety of organic semiconductors [28,30]. NiO has excellent energy-level alignment with the perovskite (CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub>) active layer. According to Kim et al. [31], when NiO is doped with copper (Cu), high-performance devices are obtained, with decent environmental stability. This is because Cu has unique electronic and structural effects and is also easy to incorporate into the solution-based fabrication of NiO [7,28,31,32]. Also, it has been noted by Chen et al. [32] that when the Cu content increases, the optical transmittance of the material reduces [32–34]. Photoluminescence (PL) spectroscopy results from the literature show that the PL quenching of Cu:NiO<sub>x</sub> is comparable to that of PEDOT:PSS [31]. These findings motivated the investigation of the combination of PEDOT:PSS and Cu:NiO<sub>x</sub> as a new composite hole transportation layer for the PSC, of which the authors are not aware of such study in the literature. A scanning electron microscopy cross-sectional view of the perovskite-based solar cell can be seen in Figure 1b.



**Figure 1.** (a) The device configuration with a PEDOT:PSS/Cu:NiO<sub>x</sub> composite hole transportation layer (HTL); (b) crosssectional view of the perovskite-based solar cell with a scanning electron microscope; (c) energy-level diagram of the different components of the perovskite solar cell (PSC) device obtained from ultraviolet-visible UV spectroscopy analysis.

The energy-level diagram of this intended device is illustrated in Figure 1c. The figure shows how the PEDOT:PSS/Cu:NiO<sub>x</sub> composite hole injection layer aligns in the PSC's multilayer structure. The thinness of the Cu:NiOx and PEDOT:PSS results in band bending, leading to a composite energy-level prediction of ~5.3 eV for the PEDOT:PSS/Cu:NiOx-based hole injection layer. This results in a PEDOT:PSS/Cu:NiOx hole transport layer, enhancing hole collection and transport efficiency [7,31,35–38]. Therefore, interface recombination is prevented [39–41].

# 2. Materials and Methods

A fluorine-doped tin oxide (FTO)/glass substrate was procured (Guangdong Youxuan Technology, Guangdong, China) and washed with a detergent solution, deionized water, acetone and isopropyl alcohol (IPA) (Pharmco Products Inc, Brookfield, CT, USA). The substrate was ultraviolet-ozone (UV-ozone)-treated for 15 min. Two solutions of Cu:NiO<sub>x</sub> were prepared by varying the concentrations of Cu:NiO<sub>x</sub> from 0.1 M to 0.2 M. First, the 0.1 M Cu:NiOx precursor solution was prepared at 65 °C and stirred for 4 h from copper (II) acetate monohydrate (Oakwood Chemical, Columbia), nickel (II) acetate tetrahydrate, and 2-methoxyethanol (Alfa Aesar, Tewksbury, MA, USA). The precursor was subsequently

deposited onto the FTO substrates at 3000 rpm for 30 s in a nitrogen glovebox. The Cu:NiOx/FTO/glass structure was then annealed in air at 400 °C for 30 min to develop Cu:NiOx nanocrystal structures. A mixture of PEDOT:PSS (Sigma Aldrich, St. Louis, MO, USA) and IPA (in 1:0.05 wt%) was filtered through a 0.45 µm polytetrafluoroethylene PTFE filter (Whatman, Buckinghamshire, UK) and spun onto the Cu:NiO<sub>x</sub>/FTO/glass substrate at 4000 rpm for 60 s. The multilayer structure was annealed on a hot plate at 130 °C for 15 min. A 5% excess lead perovskite precursor of 0.95 M was prepared in a 3:1 molar ratio of methylammonium iodide (MAI) (Dyesol, USA) to lead acetate trihydrate (Pb(OAc)<sub>2</sub>·3H<sub>2</sub>O) (Alfa Aesar, Tewksbury, MA, USA), respectively. The perovskite solution was spin-coated at 4000 rpm for 200 s and thermally annealed at 80 °C for 12 min to form the CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub> crystals [4,8,42]. The various stages involved in the fabrication process are illustrated in Figure 2.



**Figure 2.** Schematic of the stages involved in copper-doped nickel oxide (Cu:NiOx) and perovskite crystallization.

PCBM (0.02 g) (Sigma-Aldrich, St. Louis, MO, USA) in 1 mL of chlorobenzene solution was deposited onto the  $CH_3NH_3PbI_3/PEDOT:PSS/Cu:NiO_x/FTO/glass$  substrate spinning at 2000 rpm for 60 s. A 90 nm thick Ag electrode was thermally evaporated onto the active device for characterization. The configuration of the control device is Ag/PCBM/CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub>/PEDOT:PSS/FTO/glass.

The devices were tested under a Xenon-lamp solar simulator (Spectral Physics, Oriel Instruments, Irvine, CA, USA) [43], with a simulated Air Mass (AM) 1.5 global solar irradiation (100 mW/cm<sup>2</sup>). The current density–voltage curves of the devices were obtained from a Keithley 2400 source meter instrument. All the films produced were subjected to scanning electron microscopy (SEM) analyses, (SEM-XL30 Environmental FEG (FEI)), spectrophotometer (Cary 5000 UV/VIS spectrometer), and a Panalytical X'Pert Pro X-ray diffractometer (Almelo, The Netherlands). The steady-state photoluminescence excitation of the perovskite layers was studied using a picosecond time-correlated single-photon counting (TCSPC) spectrofluorometer.

#### 3. Finite Element Analysis Using Abaqus/CAE

Finite element analysis methods were birthed in the early 1950s by Turner et al. [44] at the Boeing Airplane Company for calculating stiffness influence coefficients of complextype structures. In modern engineering, finite element analysis is used to solve complex structural mechanics for the structures before real field implementation. In this research, it is used to demonstrate the mechanical behavior of the device during the fabrication of the Cu:NiO<sub>x</sub> HTL when heated in air at 400 °C for 30 min.

Abaqus/CAE (Dassault Systemes, Simulia Cooperation, Providence RI, USA) was the finite element software package used to study the effect of temperature on the mechanical behavior of the annealed Cu:NiOx composite layer on the FTO-coated glass. The model partitioned in Figure 3a has an overall thickness of 1750 nm, with that of the Cu:NiO<sub>x</sub> layer (top layer) being 125 nm thick, as shown in Figure 1b. Figure 3b,c reveals the analyzed spatial displacement (deformation), strain and stress across the Cu:NiO<sub>x</sub> /FTO/glass multilayer when annealed at 400 °C. Figure 3b shows that the Cu:NiO<sub>x</sub> layer formed would not extensively deform the multilayer. The FTO layer experiences the highest strains and stresses from the heating while the glass substrate is observed to withstand most of the thermal stresses and strains achieved in Figure 3c,d [45]. This method of fabricating the Cu:NiO<sub>x</sub> HTL can also be translated onto a polymeric or flexible substrate on the condition that their thermal capacity can accommodate the high temperatures utilized here. Simulation results of the HTL on a polyethylenetherephthalate PET substrate are available in the Supplementary Materials (Figure S1) to validate this claim.



**Figure 3.** (a) Partitioned Cu:NiO<sub>x</sub>/fluorine-doped tin oxide (FTO)/glass multilayer. (b) Deformation of multilayer after annealing at 400 °C. (c) Mises stress values for the structure at 400 °C. (d) Strain contour plot on multilayers at 400 °C.

# 4. Results and Discussion

#### 4.1. Optical Characterization

The optical characteristics of the perovskite and combined hole transport layers are presented in Figure 4. Herein, the stack layers' optical absorbance properties are presented to show the influence induced by the respective HTLs in the perovskite devices [46]. This influence was evaluated by comparing the composite HTL (Cu:NiO<sub>x</sub>/PEDOT:PSS) with the control PEDOT:PSS-based devices. Based on the ultraviolet-visible (UV-Vis) spectroscopy characteristics, it can be seen that the absorbance was largely in the visible spectrum

(~400–800 nm). The absorption profiles are similar for PEDOT:PSS, 0.1 M Cu:NiO<sub>x</sub> and 0.2 M Cu:NiO<sub>x</sub>, with the 0.1 M Cu:NiO<sub>x</sub>/PEDOT:PSS showing a marginal increase in absorbance. This shows that the composite HTLs exhibited more than 80% transmittance within the visible light spectrum and are comparable to the transmittance values of PEDOT:PSS-only [30,32,33,35,38,47–54], indicating no impediment in light transmission by the use of the composite HTL. Also, the photoluminescence spectra for the devices with different concentrations of Cu:NiO<sub>x</sub> with PEDOT:PSS and the control devices were all located at ~771 nm, indicating no red shift or blue shift (See Figure S2). This shows the bandgap in the two layers did not differ and they correspond to a bandgap of 1.61 eV [55].



**Figure 4.** Absorbance spectra of perovskite on a PEDOT:PSS/Cu:NiO<sub>x</sub>/FTO/glass substrate and control device.

## 4.2. Microstructures of Layered Films of Perovskite Solar Cells

The SEM images of the layered films of the perovskite solar cells are presented in Figure 5. Figure 5a–c represents the microstructure of the perovskite films on PEDOT:PSS/Cu:NiO<sub>x</sub> and PEDOT:PSS-only HTL, respectively. Similar grain sizes and aggregated appearances can be observed in all the SEM images of the perovskite layer. This suggests that the microstructure is not altered when deposited on the composite Cu:NiO<sub>x</sub>/PEDOT:PSS layer or on the control. This makes it conducive for device fabrication since a similar crystallization level of the perovskite is maintained.

## 4.3. X-ray Diffraction Analysis of the Composite HTL Perovskite Solar Cells

Figure 6a–c presents the diffractograms (2 theta scans) of perovskite layers spin-coated onto the compact PEDOT:PSS/Cu:NiO<sub>x</sub> and the PEDOT:PSS-only-coated FTO glass using an X-ray diffractometer, while Figure S3 shows the Cu:NiO<sub>x</sub>-coated FTO glass. All the samples showed comparable x-ray diffraction XRD characteristics, with both 0.1 M and 0.2 M Cu:NiOx-based samples displaying the same level of intensity in the peaks. The XRD reflections confirm a CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub> perovskite structure with peaks assigned to a tetragonal perovskite lattice with the following unit cell parameters: a = b = 8.85 Å, c = 12.64 Å [42,56]. The peaks marked with a "#" are assigned to the FTO [42].



**Figure 5.** Scanning electron microscopy (SEM) images of (**a**) perovskite/PEDOT:PSS/FTO/glass, (**b**) perovskite/PEDOT:PSS/0.1 M Cu:NiO<sub>x</sub>/FTO/glass, (**c**) perovskite/PEDOT:PSS/0.2 M Cu:NiO<sub>x</sub>/FTO/glass.



**Figure 6.** X-ray diffraction spectra of perovskite on (**a**) PEDOT:PSS/  $0.2 \text{ M Cu:NiO}_x/\text{FTO/glass}$ ; (**b**) PEDOT:PSS/0.1 M Cu:NiO<sub>x</sub>/FTO/glass and (**c**) PEDOT:PSS/FTO/glass, respectively.

## 4.4. Performance of Fabricated Perovskite Solar Cells

The performance of perovskite solar cells (PSCs) that were fabricated using different HTLs is presented in Figure 7. The current density–voltage (J–V) curves of the devices are shown in Figure 7a with only PEDOT:PSS and PEDOT:PSS/Cu:NiO<sub>x</sub> com-

posite (with 0.1 M and 0.2 M concentrations of Cu:NiO<sub>x</sub>) HTLs. The results show that incorporating Cu:NiO<sub>x</sub> into the PSC device improves its J-V behavior, giving a large area under the curve. The increase in the open-circuit voltage  $(V_{oc})$  of the device with  $0.1 \text{ M Cu:NiO}_{x}$ /PEDOT:PSS could result from the reduction in the potential loss at the HTL/perovskite interface due to the improved energy-level alignment, as shown by Paul et al. [39,40] and Li et al. [41] who adopted the band-bending technique to eliminate recombination at the interfaces of cadmium telluride (CdTe), thin-films and copper-indiumgallium-selenide (Cu(In,Ga)Se<sub>2</sub>) solar cells with low and high gallium (Ga) compositions. The improved  $J_{sc}$  of the Cu:NiO<sub>x</sub>/PEDOT:PSS could be attributed to the increase in the electrical conductivity of the Cu:NiO<sub>x</sub>, which provides room for more charges to be extracted from the perovskite absorber layer. However, the reduction in the efficiency of the 0.2 M  $Cu:NiO_x/PEDOT:PSS$  as an HTL could be due to surface defects caused by excess copper used as the dopant, creating recombination sites for trapped charges. Compared to the PEDOT:PSS HTL devices, there were 18% and 9% increases in the  $J_{sc}$  for the 0.1 M Cu:NiO<sub>x</sub> and 0.2 M Cu:NiOx-based devices, respectively. The Jsc of the 0.2 M Cu:NiOx-based device was lower compared to that of the  $0.1 \text{ M Cu:NiO}_{x}$ , and this could be due to the increase in charge concentration which reduces the charge mobility [32]. The efficiencies of the PE-DOT:PSS with 0.1 M Cu:NiO<sub>x</sub> devices were the highest with  $\sim$ 12.8%, followed by the 0.2 M Cu:NiO<sub>x</sub> at ~10.1%, the PEDOT:PSS-only device at ~9.8% and ~11.5% for the Cu:NiO<sub>x</sub>-only device. The changes in these values are due to the copper-doped nickel oxide blend with the PEDOT:PSS. At this point, the hole collection is enhanced along with charge transport efficiency, leading to higher electrical conductivity.



**Figure 7.** Density–voltage (*J–V*) curve of fabricated devices (**a**); photoconversion efficiency (PCE) count of the three solar cells (**b**); and photodegradation: PCE over time of PEDOT:PSS and 0.1 M Cu:NiO<sub>X</sub>/PEDOT:PSS-based solar cell (**c**).

The summary of the device parameters is presented in Table 1. It is important to note here that Figure 7a shows the J-V curves for the best device fabricated from PEDOT:PSS/Cu:NiO<sub>x</sub> HTL alongside the J-V curve from the control PEDOT:PSS HTL.

The results indicate that the PEDOT:PSS/0.1 M Cu:NiO<sub>x</sub> HTL and the light absorber (CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub>) have perfect energy-level alignment leading to an improvement in their electron-blocking capabilities [28]. According to Uisik Kwon et al., such layer configurations result in the superior stability of the device in air than PEDOT:PSS HTL-only-based devices [28]. Figure 7b presents the PCE distribution measured for the different fabricated devices. The PCEs measured from all PEDOT:PSS/Cu:NiO<sub>x</sub> HTL-based devices and the PEDOT:PSS-only device show that PSCs with 0.1 M Cu:NiO<sub>x</sub> exhibit the best device performance overall.

**Table 1.** Current-voltage characteristics and performance efficiencies obtained for the different HTL structures. ( $R_s$  and  $R_{sh}$  denote series and shunt resistance, respectively).

HTL Structure	Performance				
	$J_{sc}$ (mA·cm <sup>-2</sup> )	$V_{oc}$ (V)	FF (%)	PCE (%)	PCE <sub>avg</sub> (%)
Cu:NiO <sub>x</sub> -only <sup>a</sup>	17.92	0.98	65.00	11.45	$11.05\pm0.40$
PEDOT:PSS-only <sup>b</sup>	17.66	0.83	66.40	9.80	$9.44\pm0.30$
0.2 M Cu:NiO <sub>x</sub> /PEDOT:PSS <sup>c</sup>	19.34	0.80	65.20	10.10	$10.04\pm0.06$
0.1M Cu:NiO <sub>x</sub> /PEDOT:PSS <sup>d</sup>	21.06	0.96	62.80	12.80	$12.7\pm0.1$

<sup>a</sup> Adapted from Liu et al. [57]. <sup>b</sup> 19 devices. <sup>c</sup> 16 devices. <sup>d</sup> 6 devices.

The photodegradation performance of the two types of solar cells, namely PEDOT:PSS and 0.1 M Cu:NiO<sub>x</sub>/PEDOT:PSS-based devices, was also studied. The solar cells are continuously exposed at 1 sun (AM1.5G spectral irradiance) in an inert atmosphere. The photodegradation tests of the devices shown in Figure 7c reveal that devices with PE-DOT:PSS/Cu:NiOx hole transport layers have a slower photodegradation rate than those of PSCs made with PEDOT:PSS-only as the hole transportation layer. The PEDOT:PSS-only-based solar cell exhibits only a lifetime (T<sub>80</sub>) of about 35 min and fully degrades at about 75 min to reach 0% of its initial efficiency. On the other hand, the Cu:NiOx/PEDOT:PSS-based cell showed improved performance, exhibiting a T<sub>80</sub> of at least 100 min. A look at the other PV parameters shows that the fill factor (FF), shown in Figure S4, underpins the behavior of the PCE degradation curve. While the *Jsc* showed a faster degradation trend, the *Voc* showed a very steady slow degradation trend.

The fast degradation of the PEDOT:PSS-only device could be due to its acidic [58,59] and hygroscopic [60] nature, decaying the transparent conducting oxide (FTO) and/or the photoactive layer, which is sensitive to moisture. It could also be a result of morphological inhomogeneity with rougher interfaces. Figure 5a,b shows the SEM image of perovskite/PEDOT:PSS/FTO/glass and perovskite/PEDOT:PSS/0.1 M Cu:NiO<sub>x</sub>/FTO/glass, respectively. These phenomena could explain the rapid drop in *Jsc* and FF upon continuous illumination, previously linked to a drop in charge extraction and imbalance in charge carriers' mobility [60,61]. Further work is needed to optimize the degradation behavior and comprehend the causes of this observed behavior. The degradation results presented here are only for the illustrations of the first few results; however, the concept proves that the use of Cu:NiOx/PEDOT:PSS as an HTL can improve both the PCE and stability of PSCs.

#### 5. Conclusions

In this paper, the effect of a PEDOT:PSS and copper-doped nickel oxide (Cu:NiO<sub>x</sub>) composite hole transport layer (HTL) on the performance of perovskite solar cells (PSCs) has been presented. Thin films of inverted planar perovskite solar cells were fabricated by a photoactive layer (from a solution of methylammonium lead iodide (CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub>), [6,6]-phenyl C<sub>61</sub>-butyric acid methyl ester (PCBM), and a Ag electrode). The mechanical behavior of the device during the fabrication of the Cu:NiO<sub>x</sub> HTL was modeled with finite element simulations using Abaqus/CAE. The results showed that incorporating Cu:NiO<sub>x</sub> into the PSC devices improves its J-V behavior, giving an enhanced photoconversion efficiency

(PCE) of ~12.8% from ~9.8% and ~11.5% when PEDOT:PSS-only and Cu:NiOx-only are fabricated, respectively. The J<sub>sc</sub> for the 0.1 M Cu:NiO<sub>x</sub> and 0.2 M Cu:NiO<sub>x</sub>-based devices increased by 18% and 9%, respectively, due to the increase in the electrical conductivity of the  $Cu:NiO_x$ , which provides room for more charges to be extracted from the perovskite absorber layer. The increase in the PCE was due to the copper-doped nickel oxide blend with the PEDOT:PSS, which enhanced the hole collection and charge transport efficiency leading to higher electrical conductivity. The results indicate that the devices with the copper-doped nickel oxide (Cu:NiO<sub>x</sub>) composite hole transport layer (HTL) were slower to degrade compared with the PEDOT:PSS-only-based HTL. The increase in the open-circuit voltage ( $V_{oc}$ ) of the device with 0.1 M Cu:NiO<sub>x</sub>/PEDOT:PSS was as a result of the reduction in the potential loss at the HTL/perovskite interface due to the improved energy-level alignment using the band-bending technique. The finite element analyses showed that the  $Cu:NiO_x$  layer does not extensively deform the multilayer and the highest strains and stresses from the heating, while the glass substrate was observed to withstand the thermal stresses and strains. It can be deduced that low-temperature solution-processed PSCs with copper-doped nickel oxide composite HTL is a promising technique for enhancing the performance of perovskite solar cells.

**Supplementary Materials:** The following are available online at https://www.mdpi.com/article/10.3390/en14071949/s1, Figure S1: Simulation results of the HTL on a polyethylene terephthalate (PET), Figure S2: Photoluminescence spectra, Figure S3: X-ray diffraction spectrum of Cu:NiOx crystallographic formation on FTO coated glass, Figure S4: FF evolution over time of PEDOT:PSS and Cu:NiOx/PEDOT:PSS-based solar cells.

**Author Contributions:** J.A., D.M.S. and B.A.-T. conceptualized the work; D.M.S., A.S.Y. and J.A. performed the experiments; the simulation work was done by E.A. and J.A. and validated by O.K.O. and B.A.-T.; the analysis of data was done by J.A., A.S.Y., D.M.S., B.A.-T. and N.Y.D.; the original draft and preparation of the manuscript were written by J.A., D.M.S., B.A.-T. and N.Y.D.; the review and editing of the manuscript were done by J.A., A.S.Y., D.M.S., B.A.-T. O.K.O. and N.Y.D. The authors have given approval for the submission and publication of this manuscript in MDPI energies journal. All authors have read and agreed to the published version of the manuscript.

**Funding:** This project was funded by the BANGA-Africa programme, University of Ghana with funding from the Carnegie Corporation of New York.

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not Applicable.

Data Availability Statement: Not applicable.

Acknowledgments: The authors acknowledge Terry Alford for providing the laboratory space and collaborative technical assistance. The Macro Technology Works, Engineering Research Center, and the LeRoy Eyring Center for Solid State Science at Arizona State University, Tempe, USA are also appreciated for their useful technical support. J. Asare and D. Sanni also acknowledge the support of the Pan African Materials Institute (PAMI) of the African University of Science and Technology (AUST), Abuja, Nigeria.

Conflicts of Interest: The authors declare no conflict of interest.

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