Numerical Simulation of Sand Casting of Stainless Steel Pump Impeller

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Abstract: This paper investigates the casting defects of a stainless steel pump impeller manufactured through the sand casting process. The material characterization of austenitic steel AISI 316L was initially carried out, which examined the chemical composition of the casting and its microstructure. The next step was to determine the cause of the casting defects using numerical simulations. The numerical simulations were performed using ProCAST software (Version 18.0). Initial results of the filling and solidification simulations were conducted using the parameters employed in the actual casting process, revealing casting defects in corresponding locations. The casting process was subsequently modified to achieve improved results. This involved reconstructing the gating system, redesigning the riser, and incorporating a cylindrical chiller. The results show that the modified casting process significantly reduces the occurrence of defects in the final product. The study provides useful insights into the analysis and modification of the casting process for stainless steel pump impellers produced through sand casting. The results can help improve the quality of such products and reduce production costs associated with casting defects.

Keywords: pump impeller; casting defects; numerical simulation; sand casting; austenitic stainless steel

1. Introduction

A pump impeller is a vital rotating component of a centrifugal pump, responsible for generating centrifugal force to move fluids by imparting energy to them. Its design typically comprises a central shaft surrounded by a series of vanes or blades arranged around its circumference. Fluid enters the impeller’s center and is propelled outward by the blades, resulting in increased fluid velocity and pressure [1–4]. In recent years, impellers have become indispensable components across various industries, facilitating fluid movement and enhancing efficiency in processes ranging especially in water treatment, power generation, oil and gas production, chemical and pharmaceutical industry, food and beverage production, agriculture, marine, and HVAC systems [5,6]. As the demand for high-performance impellers continues to rise, manufacturers face the challenge of optimizing manufacturing processes to meet stringent quality standards while ensuring cost-effectiveness [7,8].

The choice of materials and manufacturing techniques significantly influences impeller performance and durability. While traditional materials like cast iron, ductile iron, stainless steel, bronze, and aluminum remain prevalent, recent research highlights the growing adoption of advanced alloys and composites. These materials offer superior mechanical properties, such as increased strength and corrosion resistance, expanding the application scope of impellers in demanding environments [9,10]. In the dynamic landscape of industrial manufacturing, impeller production has evolved significantly, leveraging a spectrum of advanced technologies to meet diverse application demands. From traditional methods like sand casting to cutting-edge techniques such as additive manufacturing and precision
casting, impeller manufacturers have access to a wide array of tools for crafting high-quality components. However, as industries demand increasingly complex designs and tighter tolerances, the integration of hybrid manufacturing approaches emerges as a compelling solution. By combining the strengths of different methods, such as additive manufacturing with precision casting or CNC machining with traditional casting, manufacturers can optimize processes, enhance efficiency, and achieve superior impeller performance [11–13].

Traditional sand casting still remains the most widely used method for impeller manufacturing, thanks to its ability to produce large-dimension impellers with high design flexibility. The simplicity of access and the relatively low cost of mold make sand casting an attractive option for manufacturers, especially when producing impellers for applications where performance takes precedence over aesthetic considerations. This method involves creating a mold from sand, into which molten metal is poured and allowed to solidify. However, its limitations in dimensional accuracy and surface finish compared to alternative manufacturing technologies highlight the need for ongoing advancements and innovations to improve the quality and efficiency of sand casting processes, ensuring its continued relevance in the modern manufacturing landscape [14–18].

The casting process, particularly sand casting, presents various challenges that can result in defects in impeller fabrication. Mismatches between the cope and drag sections of the mold can lead to irregularities in the casting. Issues like gas porosity, caused by trapped air during solidification, and inclusions such as sand or slag, can weaken the impeller structure. Cold shut, a defect arising from incomplete fusion of molten metal, and hot tears, caused by internal stresses during solidification, are also common concerns. Furthermore, shrinkage porosity, resulting from inadequate feeding of molten metal into the mold, and misruns, which occur when the metal fails to completely fill the mold cavity, can compromise the integrity of the impeller. Cold laps, caused by insufficient fluidity of the metal, and warpage, stemming from uneven cooling or internal stresses, are additional defects that can affect the impeller's performance. To mitigate these issues, manufacturers must adhere to stringent quality control measures throughout the casting process. Proper mold design, optimal pouring temperature, and precise control of cooling rates are essential factors in preventing defects. Additionally, continuous monitoring and inspection during production help identify and address any issues promptly, ensuring the production of high-quality impellers that meet performance standards [19–23].

To improve the quality and efficiency of sand casting for impellers, recent advancements have focused on the development of new and improved processes. Vacuum-assisted sand casting and investment casting are promising methods that enhance surface finish and dimensional accuracy while reducing the occurrence of defects [24–26]. Innovations in sand casting coating technologies have had a significant impact on improving the quality of impellers. These coatings act as crucial barriers between the molten metal and the mold surface, serving multifaceted roles that significantly improve the casting process. By effectively preventing erosion, they ensure the longevity of molds and cores, consequently reducing maintenance costs and enhancing operational efficiency. Coatings are important in controlling the flow rate of molten metal, optimizing solidification conditions, and minimizing the occurrence of defects. The latest advancements in coating formulations focus on harnessing nanotechnology and nanomaterials to improve adhesion and durability, thereby ensuring consistent and reliable results. Additionally, there is a growing emphasis on developing eco-friendly coatings that align with sustainability goals, minimizing environmental impact without compromising performance. Furthermore, the integration of digital technologies enables real-time monitoring and control of coating thickness, ensuring uniform application and adherence to strict quality standards [27–31].

Simulation and modeling techniques are increasingly vital for optimizing sand casting in impeller manufacturing. Finite element analysis (FEA) serves as a powerful tool for predicting deformation and stress distribution during casting, enabling engineers to identify weak points and potential defects in impeller designs. This information guides targeted design improvements, leading to enhanced impeller functionality and structural in-
On the other hand, computational fluid dynamics (CFD) plays a crucial role in simulating the flow of molten metal through the mold during casting. By visualizing and analyzing the filling and solidification stages, CFD optimizes gating and feeding systems, thereby reducing casting defects such as shrinkage porosity and ensuring a complete and uniform mold fill. These advanced simulation techniques provide a virtual testing ground for impeller designs, minimizing the need for costly and time-consuming physical prototypes [36–38]. Through iterative modeling and optimization, engineers can fine-tune the casting process, resulting in high-quality impellers with improved performance and reliability that meet modern industry demands [39,40]. Continuing advancements in simulation and modeling techniques aim to further enhance the accuracy, efficiency, and versatility of sand casting simulations for impeller manufacturing. One area of focus is the development of multiphysics simulations that integrate fluid flow, heat transfer, and structural mechanics to provide a more comprehensive understanding of the casting process. Efforts are also underway to incorporate material modeling and microstructure prediction into casting simulations, allowing for more accurate predictions of mechanical properties and performance characteristics. Furthermore, advancements in computational hardware and software are enabling the simulation of larger and more complex impeller geometries with higher fidelity and resolution. In addition, there is growing interest in leveraging artificial intelligence (AI) and machine learning techniques to enhance the predictive capabilities of casting simulations [41–43].

Considering these advancements and challenges in sand casting, this paper aims to identify causes of casting defects in a sand-casted stainless steel impeller pump and proposes an improvement of the casting process, i.e., a solution for obtaining a high-quality casting without porosity by reconstructing the gating and feeding system. Casting results were obtained through numerical simulations. Numerical simulations are useful because they allow the testing of different casting variants without the need for real casting. In casting simulations, it is possible to predict the flow of molten metal, temperature distribution, the correctness of the parameters and design of the gating and feeding system, the solidification time of the casting, and possible casting defects. The main benefits of using casting simulation are time and cost savings, increased safety, process optimization, higher efficiency, and better product quality [44–46].

2. Materials and Methods

This scientific paper focuses on analyzing a pump impeller constructed from AISI 316L austenitic stainless steel. AISI 316L stands out as the preferred material for pump impellers owing to its exceptional resistance to corrosion, particularly in aggressive environments such as marine and chloride-laden fluids. The inherent qualities of AISI 316L, including longevity, high-temperature resilience, and ease of hygiene maintenance, make it an optimal choice for demanding applications in industries like pharmaceuticals and food processing. Notably, the impeller was produced through the sand casting process, a method frequently employed in pump manufacturing due to its cost-effectiveness and suitability for intricate designs. Sand casting enables the production of complex impeller shapes at a lower cost, striking a practical balance between efficiency and affordability in the manufacturing process. Figure 1 shows the pump impeller after removal of the gating system, providing a visual reference for the discussed manufacturing method.

Upon visual inspection, the casting revealed no defects. Subsequent machining, specifically internal turning, exposed cavities at the entry points of the pump impeller, indicating product damage. This defect was identified as shrinkage porosity, as shown in Figure 2. Shrinkage porosity occurs due to the shrinkage of metal during the solidification process, leading to incomplete structure of certain areas in the casting.
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Identifying the precise cause of casting defects, particularly intricate issues like shrinkage porosity, is especially challenging in practical applications. Therefore, conducting a detailed material characterization becomes imperative. The initial step involves examining the chemical composition of the casting to confirm its alignment with the specifications for AISI 316L austenitic stainless steel. This analysis utilized the optical emission spectrometer GDS 850A, LECO (St. Joseph, MI, USA), as depicted in Figure 3.
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The chemical composition was tested and deviations were found for some chemical elements. Cr, Ni, and Mo levels are slightly lower than required by the standard for AISI 316L steel (Table 1). P and S levels are slightly higher and the Si level is higher than required. The reduced chromium content in the casting can lead to a reduction in corrosion resistance, as chromium is a critical element for forming a protective oxide layer. Chromium is an alpha-forming element that expands the range of ferrite stability, influencing the microstructure and potentially affecting mechanical properties. Conversely, nickel and manganese are highly gamma-forming elements, imparting stability and broadening the austenitic phase field. Nickel contributes to corrosion resistance and enhances ductility, while manganese can impact hardenability.

<table>
<thead>
<tr>
<th>Chemical Composition (%)</th>
<th>C</th>
<th>Si</th>
<th>Mn</th>
<th>Cr</th>
<th>Mo</th>
<th>Ni</th>
<th>N</th>
<th>P</th>
<th>S</th>
<th>Fe</th>
</tr>
</thead>
<tbody>
<tr>
<td>Casting</td>
<td>0.01</td>
<td>1.31</td>
<td>0.47</td>
<td>15.20</td>
<td>1.90</td>
<td>9.64</td>
<td>0.09</td>
<td>0.048</td>
<td>0.016</td>
<td>Balance</td>
</tr>
<tr>
<td>AISI 316L</td>
<td>≤0.03</td>
<td>≤1.00</td>
<td>≤2.00</td>
<td>16–18</td>
<td>2–3</td>
<td>10–14</td>
<td>≤0.1</td>
<td>≤0.045</td>
<td>≤0.03</td>
<td>Balance</td>
</tr>
</tbody>
</table>

After chemical characterization, the proper design of the gating system and the feeding system were analyzed and the casting parameters were checked. All dimensions of the pump impeller were precisely measured and defined before being incorporated into a designed CAD model using SolidWorks software (2020 SP3). The resulting model is shown in Figure 4, providing comprehensive isometric, front, and top views of the pump impeller along with corresponding dimensions.

The gating and feeding systems were also designed identically. In this way, an identical shape was achieved as in the real experiment. The gating system consisted of a pouring basin, sprue, sprue base, two runners, and two ingates, while the feeding system, represented by the riser, was positioned at the top of the pump impeller shaft. The vertical distance between the molten metal level in the pouring basin and the ingates level was 370 mm. The cross section of the bottom of the sprue was 315 mm², the runner was 2 × 270 mm², and the ingates were 2 × 150 mm². The gating and feeding system for the impeller pump in the form of a CAD model with labeled parts is shown in Figure 5.

Figure 3. Testing the chemical composition of the sample with the optical emission spectrometer GDS 850A, LECO (Reprinted with permission from ref. [47], 2024 LECO Corporation).
Figure 4. CAD model of a pump impeller in isometric view, front view, and top view.

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Figure 5. CAD model of a pump impeller with associated gating system and feeding system.

The CAD model was imported into the ProCAST casting simulation software. The mesh was generated by the finite element method with the densest mesh on the casting (pump impeller), followed by the gating system then the rise and core, and finally the mold with the rarest generated mesh. Figure 6 shows the distribution of the generated mesh without molds. The final number of 2D and 3D elements was 2,028,526.
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Figure 6. The distribution of elements on the CAD model (Green is a casting and blue is the core).

All material data and casting parameters were entered in the software ProCAST as used in the real casting process. The mold was made of green sand and the core was made using furan resin sand in a No-bake process. The mold and the core were coated with a water-based coating with a coating thickness of 100 µm. The molten metal was AISI 316L stainless steel. Its liquidus temperature is 1420 °C, while the solidus temperature is 1197 °C. The pouring temperature in sand casting with complex geometry is usually 100–150 °C higher than the liquidus temperature of the molten metal, and in this process, it was 1550 °C. Thermal conductivity varies depending on temperature and was 26 W/mK at pouring temperature and 14 W/mK at room temperature. The latent heat of fusion was 273 kJ/kg. The mass of the molten metal that needed to be poured into the mold was 12.5 kg. The ProCAST software automatically recognizes the contact of surfaces with different thermal conductivity. Accordingly, it was necessary to define heat transfer compound (HTC). Table 2 shows HTC amounts for different interfaces in contact.
The boundary conditions that were defined were heat dissipation and fluid flow velocity. Heat dissipation was defined by cooling the molten metal in the mold to room temperature through natural airflow. The pouring time was approximately 12 s, leading to a pouring velocity of 608 mm/s. For the fluid velocity, it was necessary to define the point and diameter of the inlet, marked with a red circle in Figure 7. Further, by inserting the pouring time which was approximately amount of 12 s, the software calculated the pouring velocity, which was 608 mm/s.

![Figure 7](image-url)

**Figure 7.** Spot and diameter of inlet of molten metal (Red color is indicated inlet).

The *k*-ε model of turbulence was used to simulate the molten metal filling into the mold and in calculating was approached with two equations. The first was the conservation equation for turbulent kinetic energy and it was described in the following expression [49]:

\[
\frac{\partial (\rho k)}{\partial t} + u_j \frac{\partial (\rho k)}{\partial x_j} - \frac{\partial}{\partial x_j} \left( \mu_T \frac{\partial k}{\partial x_j} \right) - \mu_T G - \rho \varepsilon = 0
\]  

(1)

where are *k* and *G* (turbulence generation rate):

\[
k = \frac{1}{2} \left( u^2 + v^2 + w^2 \right)
\]  

(2)

\[
G = \left( \frac{\partial u_i}{\partial x_j} + \frac{\partial u_j}{\partial x_i} \right) \frac{\partial u_i}{\partial x_j}
\]  

(3)

### Table 2. The amount of HTC at contacts of different interfaces.

<table>
<thead>
<tr>
<th>Interfaces in Contact</th>
<th>HTC (W/m²K)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Molten metal/mold</td>
<td>400</td>
</tr>
<tr>
<td>Molten metal/core</td>
<td>500</td>
</tr>
<tr>
<td>Mold/core</td>
<td>100</td>
</tr>
</tbody>
</table>

The ProCAST software automatically recognizes the contact of surfaces with different thermal conductivity. Accordingly, it was necessary to define heat transfer compound (HTC). Table 2 shows HTC amounts for different interfaces in contact.
The second was the turbulence dissipation rate and it was described in the following expression [49]:

$$\frac{\partial (\rho \varepsilon)}{\partial t} + u_j \frac{\partial (\rho \varepsilon)}{\partial x_j} - \frac{\partial}{\partial x_j} \left( \mu_T \frac{\partial \varepsilon}{\partial x_j} \right) - \frac{\varepsilon}{k} (C_1 \mu_T G - C_2 \rho \varepsilon)$$ (4)

where $\mu_T$ (turbulent eddy viscosity) is:

$$\mu_T = \frac{C_\mu \rho k^2}{\varepsilon}$$ (5)

while the default values of constants are in Table 3:

<table>
<thead>
<tr>
<th>$\sigma_k$</th>
<th>$\sigma_\varepsilon$</th>
<th>$C_1$</th>
<th>$C_2$</th>
<th>$C_\mu$</th>
</tr>
</thead>
<tbody>
<tr>
<td>1.0</td>
<td>1.3</td>
<td>1.44</td>
<td>1.92</td>
<td>0.09</td>
</tr>
</tbody>
</table>

The solidification of the molten metal was calculated with the Johnson–Mehl approximation, assuming that the solid/liquid interface is weighted by a factor of $(1 - f_s)$ to account for collision grains towards the later part of solidification [49]:

$$\frac{df_s}{dt} = (1 - f_s) \frac{4 \pi R_e^2}{R_e} N \frac{dR_e}{dt}$$ (6)

where $R_e$ is the radius of the eutectic grain and $N$ is the substrate density. When the instantaneous nucleation model is used, $N$ becomes a function of the cooling rate. For continuous nucleation, $N$ becomes a function of temperature.

3. Experimental Results

For evaluation and control of ferrite in the microstructure of stainless steel castings, the Schoefer diagram is used as a standard. The calculation requires the ratio of chromium ($Cr_e$) and nickel ($Ni_e$) equivalents, which are calculated according to the following expressions [48]:

$$Cr_e = \% Cr + 1.5 (\% Si) + 1.4 (\% Mo) + (\% Nb) - 4.99 = 14.835$$ (7)

$$Ni_e = \% Ni + 30 (\% C) + 0.5 (\% Mn) + 26 (\% N - 0.02) + 2.11 = 14.105$$ (8)

The Schoefer diagram, constructed with the calculated chromium and nickel equivalent composition ratio, is depicted in Figure 8. The $Cr_e/Ni_e$ composition ratio is 1.05, which results in a Ni content of approximately 5% in the microstructure.

The accuracy of the Schoefer diagram was determined using a ferritometer. Figure 9 displays the testing of ferrite content in the austenitic microstructure with device Ferritgehaltmesser 1.054. The device measured a ferrite content of 5%, thereby validating the results obtained in the Schoefer diagram.

The higher presence of ferrite in the austenitic microstructure results in increased shrinkage porosity due to its greater volumetric contraction during solidification, leading to the formation of internal voids within the material. On the other hand, the presence of ferrite in the austenitic microstructure can improve weldability and magnetic permeability, but it is usually avoided because it reduces the corrosion resistance of the material, increases its sensitivity to wear, and reduces its mechanical properties such as strength and hardness.

The subsequent step focused on investigating the material’s microstructure at the site of shrinkage porosity. Figure 10 shows a section of the pump impeller chosen for detailed microstructure analysis, providing visual insights into the observed porosity issues. This selected segment was cut using a band saw.
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Fluctuations in the austenitic matrix are a common occurrence in cast stainless steels, influencing properties, tribological properties, and magnetic behavior. However, increased ferrite content in the matrix increases its sensitivity to wear and reduces its mechanical properties such as strength, but it is usually avoided because it reduces the corrosion resistance of the material. The presence of ferrite in the austenitic microstructure can improve weldability and magnetic permeability, but the formation of internal voids within the material is a concern. On the other hand, the presence of porosity due to its greater volumetric contraction during solidification is a significant factor in reducing the material's mechanical integrity.

The subsequent step focused on investigating the material’s microstructure at the site where shrinkage porosity occurred. The specimen for microstructural analysis was mounted in the container, and the testing of ferrite content in the austenitic microstructure was performed using a ferritometer. The accuracy of the Schoefer diagram was determined using a ferritometer. The results obtained in the Schaefer diagram were validated by a haltermesser 1.054 device, which measured a ferrite content of 5%.

The pump impeller material displayed the testing of ferrite content in the austenitic microstructure, with the device Ferritgeometer 1.054 being used. The microstructure was analyzed using the light microscope OLYMPUS GX51F-5, which showed a distinctive two-phase microstructure consisting of an austenite matrix and the ferrite phase in the form of black lines. The austenitic matrix, characterized by a face-centered cubic (FCC) crystal structure, coexists with the ferrite phase as a result of the solidification process. The coexistence of these phases provides insights into the observed porosity issues.

Figure 8. Ferrite content, expressed in the Schoefer diagram [48].

Figure 9. Measurement of the content of ferrite in the austenite microstructure of the pump impeller.

Figure 10. Pump impeller part for microstructure analysis at the site where shrinkage porosity occurred.
After the cutting, the specimen for microstructural analysis was mounted in the conductive resin, ground, polished, and electrochemically etched in the 10% oxalic acid at a voltage of 12 V for three minutes. The microstructure was analyzed using the light microscope OLYMPUS GX51F-5 with the attached Olympus DP-25 CCD camera.

Figure 11 shows the distinctive two-phase microstructure of the impeller material, which consists of an austenite matrix and the ferrite phase in the form of black lines. The austenitic matrix, characterized by a face-centered cubic (FCC) crystal lattice, imparts corrosion resistance and high-temperature stability. The discernible ferrite phase, observed as dark lines within the microstructure, typically exhibits a body-centered cubic (BCC) crystal structure. The coexistence of these phases is a result of the solidification process during casting, influenced by factors such as alloy composition, cooling rates, and solidification conditions. The segregation of ferrite along specific boundaries or areas within the austenitic matrix is a common occurrence in cast stainless steels, influencing mechanical properties, tribological properties, and magnetic behavior.

![Microstructure of the casted AISI 316L material.](image)

**Figure 11.** Microstructure of the casted AISI 316L material.

Microhardness was measured using the Vickers method HV 0.1 with a test load of 0.9807 N. It was measured on both phases separately: in the region where the austenite phase is dominant, and in the region where the ferrite phase is dominant.

Figure 12 shows the resulting indentations in the predominantly austenitic phase (a) and in the predominantly ferritic phase (b) of the sample.

![Microhardness measurement indentations: (a) predominantly austenitic phase; (b) predominantly ferritic phase.](image)

**Figure 12.** Microhardness measurement indentations: (a) predominantly austenitic phase; (b) predominantly ferritic phase.
The pronounced contrast in microhardness values between the predominantly austenitic and ferritic phases highlights the mechanical distinctions within the dual-phase microstructure. The elevated average microhardness of 133 HV0.1 in the predominantly austenitic region consistently reflects the heightened strength and hardness commonly associated with austenitic stainless steels. In contrast, the diminished average microhardness of 96 HV0.1 in the predominantly ferritic phase is in line with the anticipated lower hardness characteristic of ferritic structures.

4. Simulation Results

Simulation results for the casting of the pump impeller were obtained using ProCAST software. The choice of ProCAST was based on its user-friendly interface, high reliability, and capability to provide realistic representations of the casting process. Figure 13 shows the filling of the mold cavity with molten metal at different stages. It was observed that the level of molten metal in the sprue was lower than the level of molten metal in the mold cavity. This discrepancy may suggest that the gating system was oversized. Such a condition could lead to uneven distribution of metal in the mold cavity, potentially resulting in variations in wall thickness and other casting defects.

The next analysis concerns the solidification process of the molten metal in the mold cavity. The critical areas in the mold cavity are those that solidify last, so the feeding systems were arranged to compensate for volume loss. Thin sections in contact with the mold should solidify first, then thicker sections and the riser at the end, which is called directional solidification. In Figure 14, it can be seen that the directional solidification towards the riser was not achieved. Since the lower side of the pump impeller was no longer fed by the riser, the possibility of casting defects was expected in this area.

In the ProCAST software, it is possible to estimate the probability of casting defects. Figure 15 shows the probability of shrinkage porosity, revealing its occurrence at the point where the melt enters the pump impeller. The analysis indicated that the probability of casting defects in this area exceeded 60%. Consequently, numerical simulations were found to accurately predict casting defects at the same location observed in the actual casting process.
Figure 14. Solidification process in two different stages.

It can be concluded that the potential cause of shrinkage porosity at the entrance of the pump impeller is the uneven wall thicknesses of the casting. The walls of the blades are significantly thinner than the rest of the casting, impeding the compensation of shrinkage from the riser during the solidification of the casting. Given the constraint of not being able to modify the casting design, potential solutions may involve optimizing the gating system or incorporating additional elements to facilitate directional solidification and eliminate the observed shrinkage porosity.
5. Optimisation

Taking into account the material characterization and simulation results, proposed improvements included the usage of the chill, reconstruction of the gating system, and redesigning of the riser. By placing the cylindrical chill at the bottom of the impeller, the solidification process can be better directed to the riser, thus reducing the possibility of casting defects in this area. The chill was made of cast iron. Its thermal conductivity at room temperature is 80.5 W/mK, and it decreases at higher temperatures. The interface chill is in contact with the mold and the molten metal, so the following HTC values are defined, as shown in Table 4.

Table 4. The amount of HTC between chill and mold and between chill and molten metal.

<table>
<thead>
<tr>
<th>Interfaces in Contact</th>
<th>HTC (W/m²K)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Chill/mold</td>
<td>500</td>
</tr>
<tr>
<td>Chill/molten metal</td>
<td>2000</td>
</tr>
</tbody>
</table>

The cross sections of the gating system parts have been changed. The bottom of the sprue amounts to 277 mm², runners 2 × 88 mm² and ingates 2 × 66 mm². With this reconstruction, a pressurized gating system was obtained with the ingate as a critical section. The pressurized gating system provides a more efficient use of the feeder and ensures that the molten metal reaches all parts of the mold cavity, including thin-walled sections (in this study, the blades). The riser contained a cylindrical shape, and the difference was manifested by a larger cross section and reduced height. This decreased the vertical distance between the level of the molten metal in the pouring base and the level of the ingates, and it now amounts to 276 mm. Figure 16 shows the casting with the pressurized gating system and the feeding system with the additional cylindrical chill, which is marked in green.

The results of the impeller casting with the new gating system design were obtained by numerical simulations. The first simulation performed was the solidification process. Figure 17 shows the directional solidification of the casting towards the riser. Molten metal located after ingates was trapped leading to shrinkage porosity in this area. The casting was satisfactory because that part would be removed through machine cutting.
**Figure 16.** CAD model of a pump impeller with improved gating and feeding system and additional chill.

**Figure 17.** Solidification process in two different stages in the improved version of gating system.
Further analysis was performed to determine the occurrence of shrinkage porosity in the casting. Figure 18 shows the parts of the casting where shrinkage porosity occurred. This time, the shrinkage porosity did not occur at the entrance of the pump impeller, which was a great improvement. The lower part of the shaft indicates the presence of porosity, but it has already been mentioned that this area is not part of the casting. The results of the numerical simulation show that with this improvement it was possible to obtain a high-quality casting without casting defects.

![Figure 18. Locations of the shrinkage porosity in the gating system with a probability of more than 60% in the improved version.](image)

6. Discussions

The results of this study provide valuable insights for the analysis and modification of the casting process for stainless steel pump impellers produced via the sand casting method. It requires careful consideration of the materials used, the design of the gating and feeding systems, and the optimum selection of parameters to obtain a high-quality product without casting defects.

The simulations revealed that the casting defects were caused by earlier solidification of the thin-walled blades of the pump impeller, which prevented further supply of molten metal from the riser to the entrance of the pump impeller where shrinkage porosity occurred.

Consequently, the following changes were proposed for improvement:

- Reconstruction of the pressurized gating system;
- Redesigning of the riser;
- Placement of a cylindrical chill in the lower part of the pump impeller.

The proposed changes were tested through numerical simulations using ProCAST software. The results showed that the occurrence of casting defects in the casting was eliminated, thus improving the quality of the final product. The improved gating system, modified riser, and added cylindrical chill effectively directed the solidification to the riser and prevented the formation of shrinkage porosity in the lower part of the pump impeller. However, further research is needed to validate the numerical simulations with actual casting experiments.
7. Conclusions

The process of impeller casting represents a highly intricate procedure due to the complex geometry of the casting, where even minor oversights can result in product defects. To optimize casting processes, reduce costs, and meet the demands of today’s competitive market, numerical simulations are indispensable. Utilizing simulations provides valuable insights to anticipate potential issues or discrepancies during casting or solidification processes, aiding in both defect identification and casting parameter optimization to ensure efficient production of high-quality components. But while numerical simulations offer valuable insight into the casting process, their accuracy depends on a variety of factors, including the following:

- Material properties and behavior in different conditions;
- Boundary conditions and environmental factors during casting;
- Interaction between molten metal and mold;
- Accuracy of the computational model and algorithms used;
- Resolution and mesh generation techniques used in the simulation.

Although simulations offer a predictive understanding of the casting process, full reliance on them, especially for complex castings like impellers, is not advisable. Therefore, further research is necessary to validate simulation results through real experiments, involving casting into a sand mold and incorporating all enhanced modifications detailed in the research paper. Each validation of numerical simulation results ensures reliability, accuracy, and credibility, enabling further improvements and the effective utilization of simulation results, particularly in the field of casting.

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